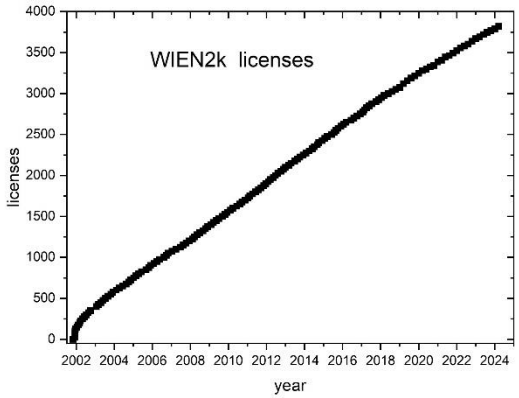
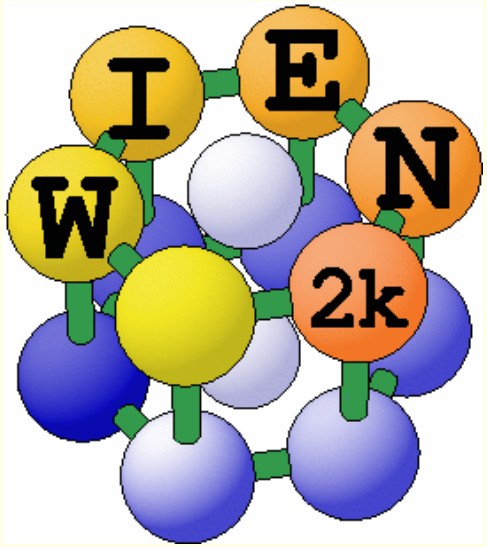


WIEN2k software package



WIEN97: ~500 users
WIEN2k: ~3900 users

An Augmented Plane Wave Plus Local Orbital Program for Calculating Crystal Properties

Peter Blaha
Karlheinz Schwarz
Georg Madsen
Dieter Kvasnicka
Joachim Luitz
Robert Laskowski
Fabien Tran
Laurence Marks

November 2001
Vienna, AUSTRIA
Vienna University of Technology

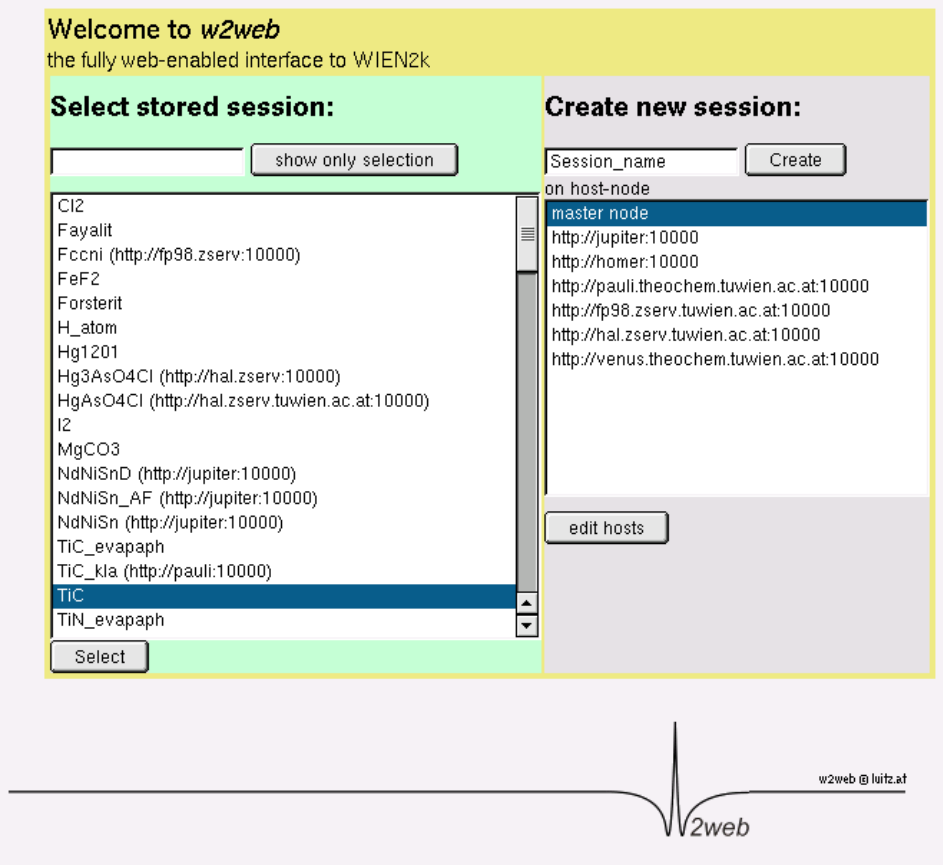
http://www.wien2k.at

General remarks on WIEN2k

- WIEN2k consists of many independent F90 programs, which are linked together via C-shell scripts.
- Each „case" runs in his own directory ./case
- The „master input" is called case.struct
- Initialize a calculation: init_lapw
- Run scf-cycle: run_lapw (runsp_lapw)
- You can run WIEN2k using any www-browser and the w2web interface, but also at the command line in an xterm.
- Input/output/scf files have endings as the corresponding programs:
  - *case.output1…lapw1; case.in2…lapw2; case.scf0…lapw0*
- Inputs are generated using STRUCTGEN(w2web) or makestruct and init_lapw

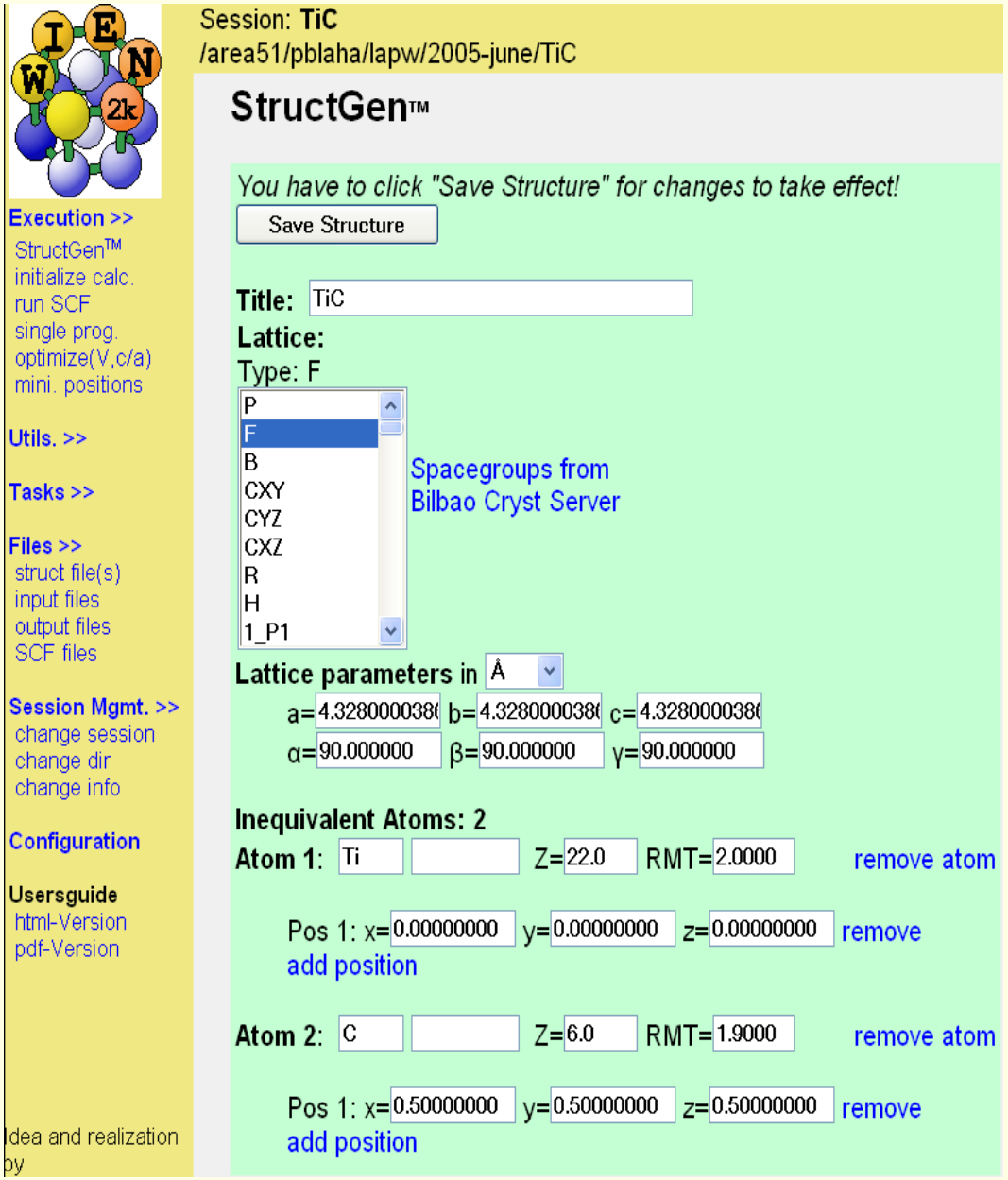
w2web: the web-based GUI of WIEN2k

- Based on www
  - *WIEN2k can be managed remotely via w2web*
- Important steps:
  - *start w2web on all your hosts*
    - login to the desired host (ssh)
    - w2web (at first startup you will be asked for username/password, port-number, (master-)hostname. creates ~/.w2web directory)
  - *use your browser and connect to the (master) host:portnumber*
    - firefox http://fp98.zserv:10000
  - *create a new session on the desired host (or select an old one)*



w2web GUI (graphical user interface)

- Structure generator
  - *spacegroup selection*
  - *import cif or xyz file*
- step by step initialization
  - *symmetry detection*
  - *automatic input generation*
- SCF calculations
  - *Magnetism (spin-polarization)*
  - *Spin-orbit coupling*
  - *Forces (automatic geometry optimization)*
- Guided Tasks
  - *Energy band structure*
  - *DOS*
  - *Electron density*
  - *X-ray spectra*
  - *Optics*



Spacegroup $P4_2/mnm$

Structure given by:

- spacegroup
- lattice parameter
- positions of atoms (basis)

Rutile TiO_2:

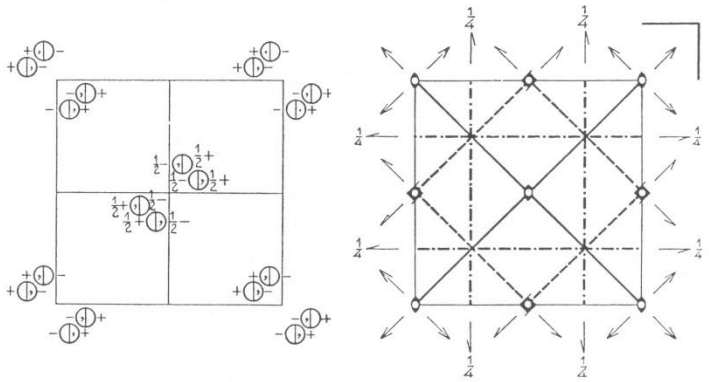
$P4_2/mnm$ (136)

a=8.68, c=5.59 bohr

Ti: (0,0,0)

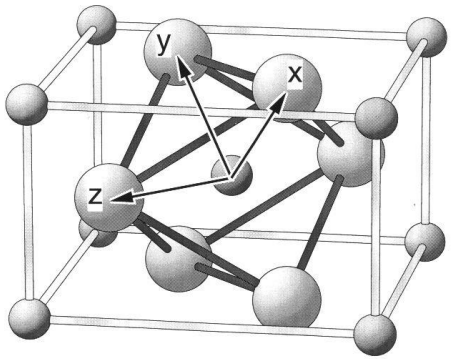
O: (0.304,0.304,0)

$P4_2/mnm$ D_{4h}^{14} No. 136 $P\,4_2/m\,2_1/n\,2/m$ $4/m\,m\,m$ Tetragonal



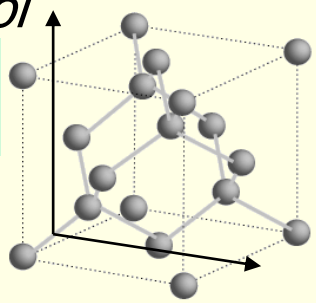
Origin at centre (*mmm*)

Number of positions, Wyckoff notation, and point symmetry			Co-ordinates of equivalent positions	Conditions limiting possible reflections
				General:
16	*k*	1	x,y,z; $\bar{x},\bar{y},z$; $\frac{1}{2}+x,\frac{1}{2}-y,\frac{1}{2}+z$; $\frac{1}{2}-x,\frac{1}{2}+y,\frac{1}{2}+z$; $x,y,\bar{z}$; $\bar{x},\bar{y},\bar{z}$; $\frac{1}{2}+x,\frac{1}{2}-y,\frac{1}{2}-z$; $\frac{1}{2}-x,\frac{1}{2}+y,\frac{1}{2}-z$; y,x,z; $\bar{y},\bar{x},z$; $\frac{1}{2}+y,\frac{1}{2}-x,\frac{1}{2}+z$; $\frac{1}{2}-y,\frac{1}{2}+x,\frac{1}{2}+z$; $y,x,\bar{z}$; $\bar{y},\bar{x},\bar{z}$; $\frac{1}{2}+y,\frac{1}{2}-x,\frac{1}{2}-z$; $\frac{1}{2}-y,\frac{1}{2}+x,\frac{1}{2}-z$.	*hkl*: No conditions *hk*0: No conditions 0*kl*: $k+l=2n$ *hhl*: No conditions
				Special: as above, plus
8	*j*	*m*	x,x,z; $\bar{x},\bar{x},z$; $\frac{1}{2}+x,\frac{1}{2}-x,\frac{1}{2}+z$; $\frac{1}{2}-x,\frac{1}{2}+x,\frac{1}{2}+z$; $x,x,\bar{z}$; $\bar{x},\bar{x},\bar{z}$; $\frac{1}{2}+x,\frac{1}{2}-x,\frac{1}{2}-z$; $\frac{1}{2}-x,\frac{1}{2}+x,\frac{1}{2}-z$.	no extra conditions
8	*i*	*m*	$x,y,0$; $\bar{x},\bar{y},0$; $\frac{1}{2}+x,\frac{1}{2}-y,\frac{1}{2}$; $\frac{1}{2}-x,\frac{1}{2}+y,\frac{1}{2}$; $y,x,0$; $\bar{y},\bar{x},0$; $\frac{1}{2}+y,\frac{1}{2}-x,\frac{1}{2}$; $\frac{1}{2}-y,\frac{1}{2}+x,\frac{1}{2}$.	
8	*h*	2	$0,\frac{1}{2},z$; $0,\frac{1}{2},\bar{z}$; $\frac{1}{2},0,\frac{1}{2}+z$; $\frac{1}{2},0,\frac{1}{2}-z$; $\frac{1}{2},0,z$; $\frac{1}{2},0,\bar{z}$; $0,\frac{1}{2},\frac{1}{2}+z$; $0,\frac{1}{2},\frac{1}{2}-z$.	*hkl*: $h+k=2n$; $l=2n$
4	*g*	*mm*	$x,\bar{x},0$; $\bar{x},x,0$; $\frac{1}{2}+x,\frac{1}{2}+x,\frac{1}{2}$; $\frac{1}{2}-x,\frac{1}{2}-x,\frac{1}{2}$.	
4	*f*	*mm*	$x,x,0$; $\bar{x},\bar{x},0$; $\frac{1}{2}+x,\frac{1}{2}-x,\frac{1}{2}$; $\frac{1}{2}-x,\frac{1}{2}+x,\frac{1}{2}$.	
4	*e*	*mm*	$0,0,z$; $0,0,\bar{z}$; $\frac{1}{2},\frac{1}{2},\frac{1}{2}+z$; $\frac{1}{2},\frac{1}{2},\frac{1}{2}-z$.	
4	*d*	$\bar{4}$	$0,\frac{1}{2},\frac{1}{4}$; $\frac{1}{2},0,\frac{1}{4}$; $0,\frac{1}{2},\frac{3}{4}$; $\frac{1}{2},0,\frac{3}{4}$.	
4	*c*	2/*m*	$0,\frac{1}{2},0$; $\frac{1}{2},0,0$; $0,\frac{1}{2},\frac{1}{2}$; $\frac{1}{2},0,\frac{1}{2}$.	
2	*b*	*mmm*	$0,0,\frac{1}{2}$; $\frac{1}{2},\frac{1}{2},0$.	
2	*a*	*mmm*	$0,0,0$; $\frac{1}{2},\frac{1}{2},\frac{1}{2}$.	



Structure generator

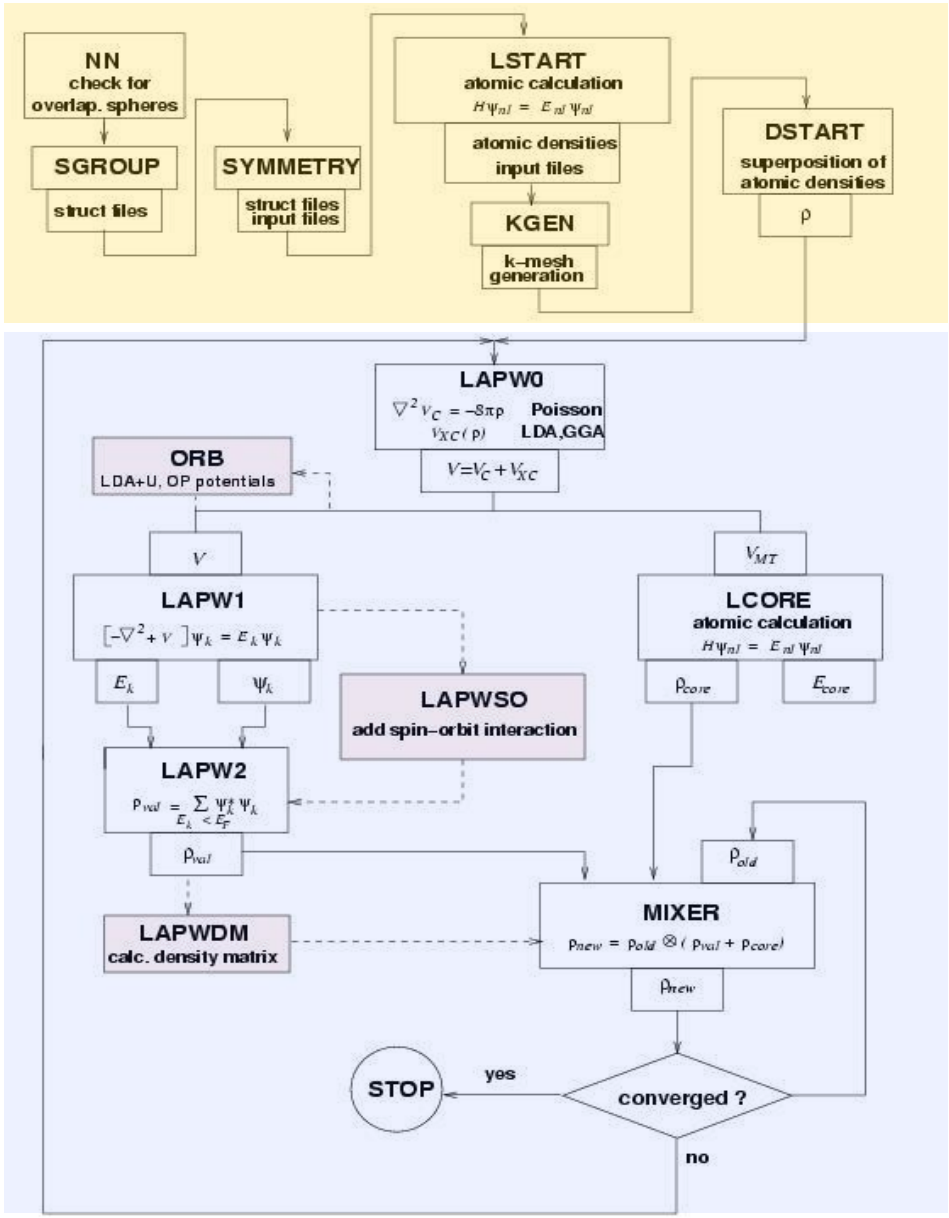
- Specify:
  - *Number of nonequivalent atoms*
  - *lattice type (P, F, B, H, CXY, CXZ, CYZ) or spacegroup symbol*
    - if existing, you must use a SG-setting with inversion symmetry:
      - Si: ±(1/8,1/8,1/8), not (0,0,0)+(1/4,1/4,1/4)!
  - *lattice parameters a,b,c (in Å or bohr)*
  - *name of atoms (Si) and fractional coordinates (position)*
    - as numbers (0.123); fractions (1/3); simple expressions (x-1/2,…)
    - in fcc (bcc) specify just one atom, not the others in (1/2,1/2,0; …)



- „save structure "
  - *updates automatically Z, r0, equivalent positions*
- „set RMT and continue": (specify proper "reduction" of NN-distances)
  - *non-overlapping „as large as possible" (saves time, may require L^{vns}=6(8))*
  - *RMT for sp (d) - elements 10-20 % smaller than for d (f) elements*
  - *largest spheres not more than 50 % larger than smallest sphere*
  - *Exception: H in C-H or O-H bonds: RMT~0.6 bohr (RKMAX~3-4)*
  - *Do not change RMT in a „series" of calculations, RMT equal for same atoms*
- „save structure – save+cleanup"

Program structure of WIEN2k

- init_lapw
  - *step-by-step or batch initialization*
  - *symmetry detection (F, I, C-centering, inversion)*
  - *input generation with recommended defaults*
  - *quality (and computing time) depends mainly on **k-mesh** and **R.Kmax** (determines #PW)*
- run_lapw
  - *scf-cycle*
  - *optional with SO and/or LDA+U*
  - *different convergence criteria (energy, charge, forces)*
- save_lapw tic_gga_100k_rk7_vol0
  - *cp case.struct and clmsum files,*
  - *mv case.scf file*
  - *rm case.broyd* files*



RKMAX (usually set by "init_lapw –prec 0-3")

- The convergence criterion in APW is the product of R_{MT}.Kmax

$$\Psi = \sum_{K_n}^{KMAX} c_{K_n} \; e^{iK_n r}$$

- http://www.wien2k.at/reg_user/faq/rkmax.html
- medium quality convergence for **smallest** atom:

Rkmax	Element
3.0	H
4.5	Li
5.0	Be, B, Si
5.5	C, P
6.0	N, S
6.5	O, Cl, Na, K, Rb, Cs, Mg, Ca, Sr, Ba, Al
7.0	F
7.5	Sc-Cr, Ga-Br, Y-Mo
8.0	Mn-Zn, Ru-Cd, In-I, La, Ce, Hf-Re
8.5	Os-At, Pr-Lu, Ac-Lr

- basis set scales with $RKmax^3$
- cputime scales with N_{PW}^3
- increasing Rkmax by 10 %
  ⇒ doubles cputime

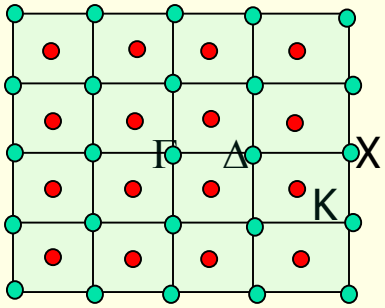
START with **SMALL** Rkmax (relaxation), **increase/test** later

BZ integration, "FERMI"-methods

- Replace the "integral" of the BZ by a finite summation on a mesh of "k-points"

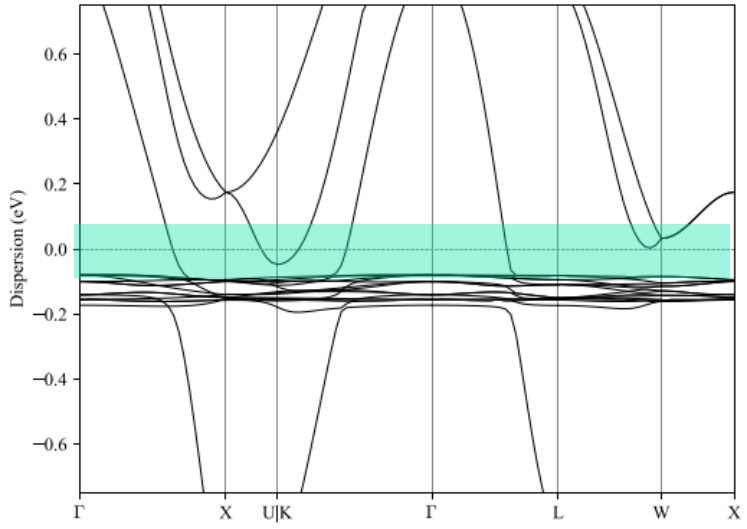
$$\rho(r) = \sum_{n}^{E_n<E_F} \int \psi^*_{k,n}\, \psi_{k,n}\, d^3k \quad = \quad \sum_{k,n} w_{k,n}\, \psi^*_k \psi_k$$

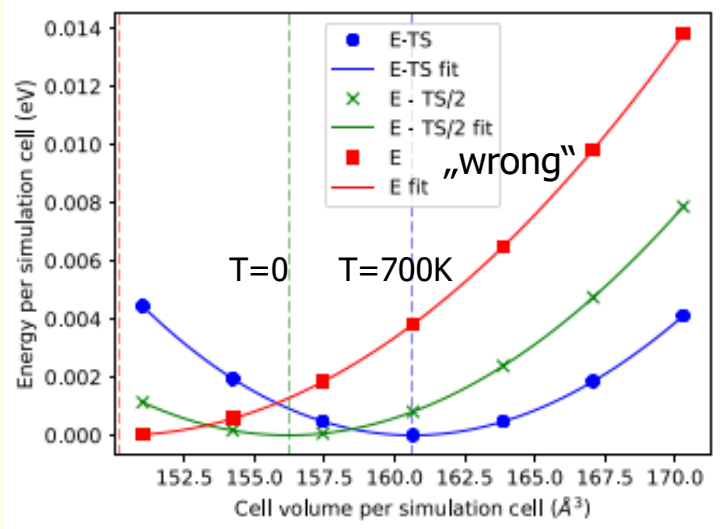
- weights $w_{k,n}$ depend on k and bandindex n (occupation)
  - *for full "bands" the weight is given by "symmetry"*
    - w(Γ)=1, w(x)=2, w(Δ)=4, w(k)=8
    - ➡ shifted "Monkhorst-Pack" mesh



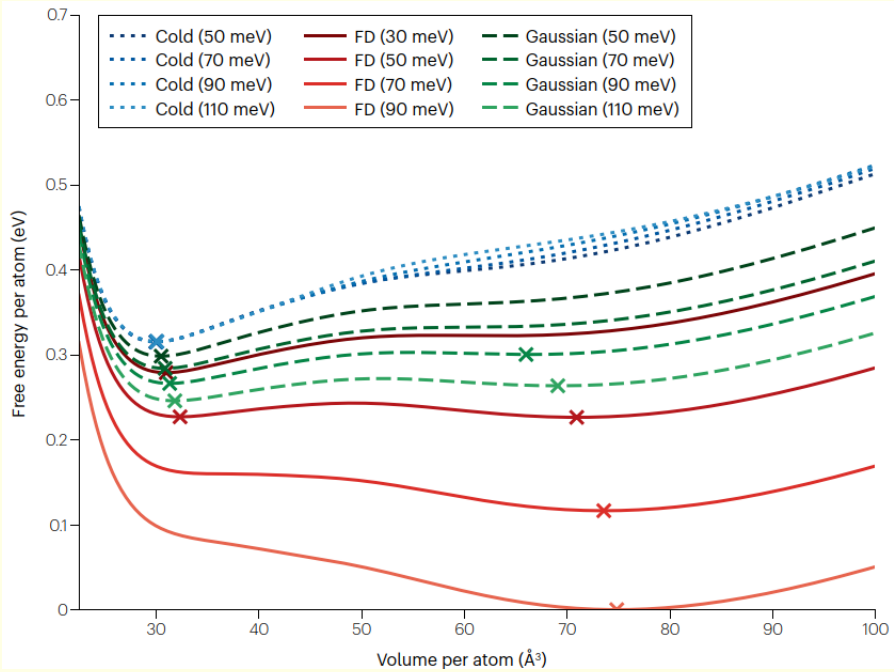
  - *for partially filled bands (metals) one must find the Fermi-energy (integration up to NE) and determine the weights for each state $E_{k,n}$*
    - linear tetrahedron method+"Bloechl" corrections (TETRA), T=0, inconsistent forces
    - "broadening methods"
      - gauss-broadening (GAUSS 0.005)
      - temperature broadening (Fermi-Dirac smearing)
        - TEMP 0.002 (-TS/2 correction, extrapolating to T=0, inconsistent forces)
        - TEMPS 0.002 (finite T (~300K) result, -TS corr., **consistent** forces)
  - *broadening useful to damp scf oszillations, but dangerous (magnetic moment)*

Er in diamond structure





T=0.0045 Ry
~ 700 K



volume may change by a factor of 2.

k-mesh generation

- x kgen (generates k-mesh and reduces to irreducible wedge using symmetry)
  - *automatically "adds inversion"*
    - time inversion holds and E(k) = E(-k)
    - except in magnetic spin-orbit calculations (`x -so kgen`; uses case.ksym file)
    - `x -fbz kgen` (generates „full mesh" in BZ)
  - *always "shift" the mesh for scf-cycle*
    - gaps often at Γ ! (might not be in your mesh)
  - *small unit cells and metals require large k-mesh (1000-100000)*
  - *large unit cells and insulators need only 1-10 k-points*
  - *use at first a fairly **coarse** mesh for scf/relaxations*
  - *continue **later** with **finer** mesh*
    - mesh was good if nothing changes and scf terminates after few (3) iterations
  - *use even finer meshes for DOS, spectra, optics,...*

x kgen

- You can specify the k-mesh by:
  - *Total number of k-points in full BZ (typically 1 - 50000)*
    - 1000 = 10x10x10 in cubic sym
    - = 13x13x5 in tetragonal with c/a=2.2
  - *3 divisions for the 3 rec.latt.vectors (no check, give correct ratio):*
    - 0
    - 4 4 4 (for a cubic insulator), or 20 20 1 (for a metallic surface slab)
  - *delta K (in bohr $^{-1}$, typically between 0.5 – 0.02)*
    - -1
    - 0.1
- **init_lapw –prec 0-3** (for metals) or **0n-3n** (for insulators) will set the k-mesh automatically according to the size of the cell.
- It will also set **ALL** other parameters (RKMAX, GMAX, L-VNS, RMT, HDLOs, FFT-mesh) according to the desired precision.

Program execution:

- All programs are executed via the „master" shell-script x_lapw

  x lapw2 –up –orb

- This generates a „def" file: lapw2.def

```
5,'tin.in2c',          'old',    'formatted'
6,'tin.output2up',   'unknown','formatted'
8,'tin.clmvalup',    'unknown','formatted'
10,'./tin.vectorup','unknown','unformatted'
```

- and executes: lapw2c lapw2.def
- All WIEN2k-shell scripts have long and short names:
  - *x_lapw; runsp_lapw, runfsm_lapw ➔ x; runsp; runfsm*
- All scripts have a „help" switch „-h", which explains flags and options (without actually execution)

  x –h *x lapw1 -h*

scf-cycle

- run_lapw [options] (for nonmagnetic cases)

-ec 0.0001	*convergence of total energy (Ry)*
-cc 0.0001	*convergence of charge distance (e^-)*
-fc 1.0	*convergence of forces (mRy/bohr)*
-it (-it1,-it2 , -noHinv)	*iterative diagonalization (large speedup)*
-p	*parallel calculation (needs .machines file)*
-so	*add spin-orbit (only after „initso")*
Spacegroups without inversion use automatically lapw1c, lapw2c (case.in1c,in2c)	

- case.scf: master output file, contains history of the scf-cycle
  - *most information is stored with some „labels" (grep :label case.scf)*

```
:ENE      :DIS      :FER  :GAP  :CTO001        :NTO001              :QTL001
:FGL002:   2.ATOM          13.767          13.767          0.000  total forces
:LAT      :VOL      :POSxxx
```

Getting help

- *_lapw –h „help switch“ of all WIEN2k-scripts
- help_lapw:
  - *opens usersguide.pdf; Use ^f keyword to search for an item („index")*
- html-version of the UG: ($WIENROOT/SRC_usersguide/usersguide.html)
- http://www.wien2k.at/reg_user
  - *FAQ page with answers to common questions*
  - *Update information: When you think the program has an error, please check newest version*
  - *Textbook section: DFT and the family of LAPW methods by S.Cottenier*
  - *Mailing-list:*
    - subscribe to the list (always use the same email)
    - full text search of the „digest“ (your questions may have been answered before)
    - posting questions: Provide sufficient information, locate your problem (case.dayfile, *.error, case.scf, case.outputX).
    - „My calculation crashed. Please help.“ This will most likely not be answered.

most common problems

- „QTL-B" value too large - STOP (or :WARN): "ghostbands"
  - *identify for which **eigenvalue**, **atom** and ℓ it happens, check E_F (case.scf2, case.output2)*
  - *identify the corresponding linearization energies in case.scf1*
  - *change the corresponding linearization energy in case.in1*
    - compare and check with :EPL and :EPH lines in case.scf2
    - default E-parameters are adapted automatically but may need changes for
      - surfaces, molecules (negative EF) or heavy elements (EF often larger than 1.0)
    - add a local orbital with $\ddot{u}$ (HDLO) (or adjust its energy)
  - *if QTL-B occurs for an atom with large RMT (and you have other atoms with small RMT), reduce RMT („numerical linear dependency")*
- scf-cycle diverges (grep :DIS case.scf):
  - *check structure (most likely a wrong structure caused divergence);*
  - *check E-parameters (see above), check :NEC01 (correct number of e^-)*
  - *rm *.broyd* case.scf; x dstart*
  - set STIFFER in case.inm

case.in1

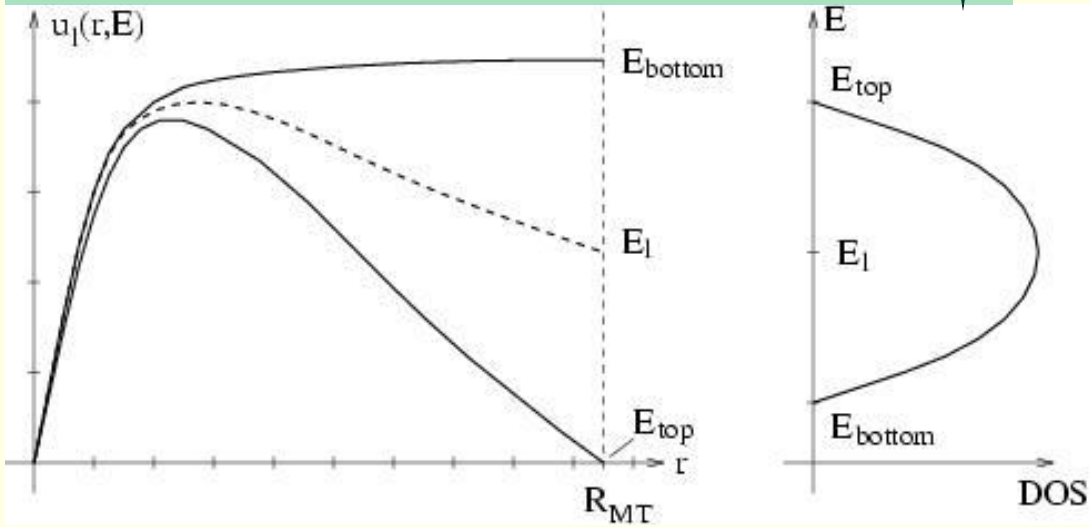
set E_l to E_F-0.2 Ry

```
WFFIL    EF=0.634                    (WFPRI, SUPWF)
  7.00     10    4        (R-MT*K-MAX; MAX L IN WF, V-NMT
  0.30      5    0        global E-param with N other, napw
  0      0.30        0.000 CONT 1        Es
  0     -3.72        0.005 STOP 1        Es-LO with search
  1     -2.07        0.010 CONT 1        Ep    with search
  1      0.30        0.000 CONT 1        Ep-LO
  2      0.30        0.010 CONT 1        0/1/2...LAPW/APW+lo/HDLO
K-VECTORS FROM UNIT:4   -7.0   1.5   16   emin/emax; nband
```

$$\Psi = \sum_{K_n}^{KMAX} c_{K_n} \, e^{iK_n r}$$

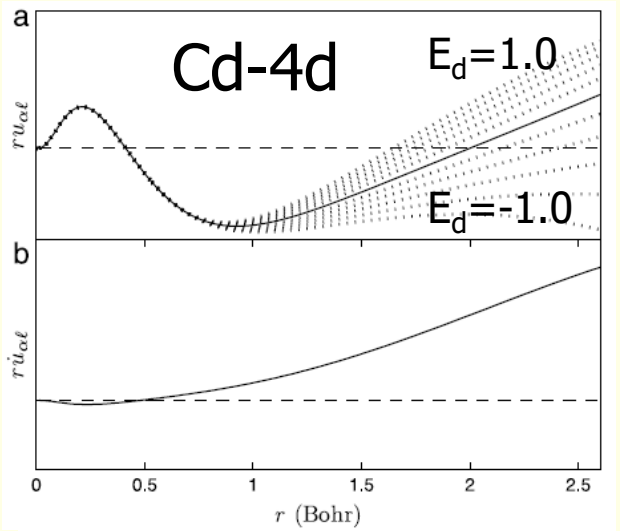
$$\Phi_{K_n} = \sum_{l}^{l\,\max} A_{lm} u_l(E_l, r) Y_{lm}$$

$$H_{n,m}^{NS} = \left\langle \Phi_l \middle| V_{LM}^{NS} \middle| \Phi_{l'} \right\rangle$$



HDLOs: case.in1 (set automatically with init –prec 2-3)

- f (d) wavefunctions have a large E-dependency in cases with large RMT
- For high precision calculations extend the basis set with a HDLO (high derivative LO):

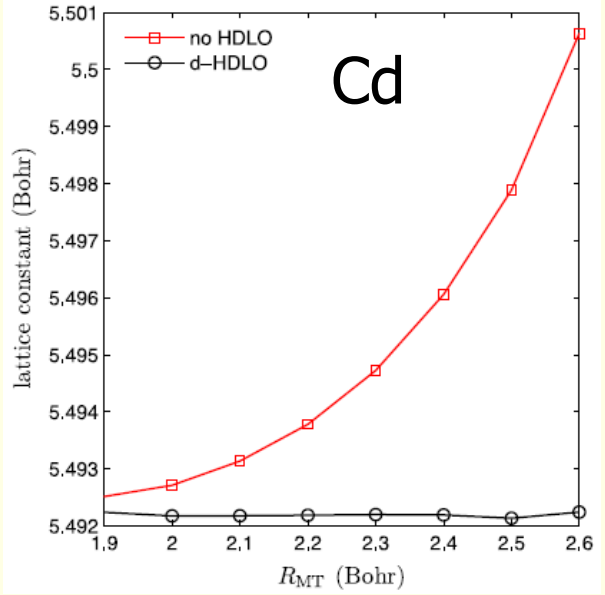


$$\Phi_{K_n} = \sum_l A_{lm}(K_n) u_l(E_l, r) Y_{lm}$$ **APW**

$$\phi_{l,atom} = (A_{lm} u_{lm} + B_{lm} \dot{u}_l) Y_{lm}$$ **lo**

$$\phi_{l,atom} = (A_{lm} u_{lm} + C_{lm} \ddot{u}_l) Y_{lm}$$ **HDLO**

- `2 0.30 0.010 CONT 1` **APW+lo**
- `2 0.30 0.010 CONT 2` **HDLO**



- **F.Karsai et al., CPC 220, 230(2017)**

case.klist, case.in2

```
GAMMA          0     0     0    40   1.0   IX, IY, IZ, IDIV, WEIGHT
               1     0     0    40   6.0
...
X             40     0     0    40   3.0
END
```

case.in2:

```
TOT                                  (TOT,FOR,QTL,EFG,FERMI)
-9.0 16.0      0.50 0.05             EMIN, NE,  ESEPARMIN, ESEPAR0
TETRA     0.000                      (GAUSS,ROOT,TEMP,TETRA,ALL    eval)
  0 0   4 0   4 4   6 0   6 4
  0 0   4 0   4 4   6 0   6 4
  14.                 GMAX(automatically larger for small spheres H)
FILE                  FILE/NOFILE  write recprlist
```

$$\rho(r) = \sum_{LM} \rho_{LM}(r) Y_{LM}(\hat{r}) \qquad \rho(r) = \sum_{G}^{GMAX} \rho_G e^{iGr}$$

Properties with WIEN2k - I

- Energy bands
  - *classification of irreducible representations*
  - *´character-plot´ (emphasize a certain band-character)*
- Density of states
  - *including partial DOS with l and m- character (eg. p_x , p_y , p_z)*
- Electron density, potential
  - *total-, valence-, difference-, spin-densities, ρ of selected states*
  - *1-D, 2D- and 3D-plots (Xcrysden)*
  - *X-ray structure factors*
  - *Bader´s atom-in-molecule analysis, critical-points, atomic basins and charges ($\nabla\rho.\vec{n}=0$)*
  - *spin+orbital magnetic moments (spin-orbit / LDA+U)*
- Hyperfine parameters
  - *hyperfine fields (contact + dipolar + orbital contribution)*
  - *Isomer shift*
  - *Electric field gradients*
  - *NMR chemical and Knight shifts*

partial charges "qtl" + DOS

- be sure to have case.vector on a dense tetrahedral mesh after a scf calculation
  - *eventually:*
    - x kgen
    - edit case.in1 (larger Emax)
    - x lapw1
- x lapw2 –qtl

  $\Psi_n^*\Psi_n=1=q_{out}+\sum_t^{at}\sum_l q_{t,l}$
- case.outputt
  - *integrated DOS*
- case.dos1ev (3ev)
  - *text-file for plotting*
  - *E-zero at E_F*

Session: **TiC**
/susi/pblaha/lapw/TiC

Density of states

x lapw2 -qtl	Calculate partial charges ☑ interactively
edit TiC.int	Edit input-file for TETRA
x tetra	Calculate partial DOS ☑ interactively
edit TiC.outputt	Check output of TETRA
dosplot	Plot DOS

Session: **TiC**
/susi/pblaha/lapw/TiC

File:
/susi/pblaha/lapw/TiC/TiC.int

contine with DOS | Save | Download this file:

Header from TiC.qtl:

```
ATOM   1  tot,0,1,2,3,xdos(i,j),j=1,i),i=1,lxdos2)
ATOM   2  tot,0,1,2,D-eg,D-t2g,3
```

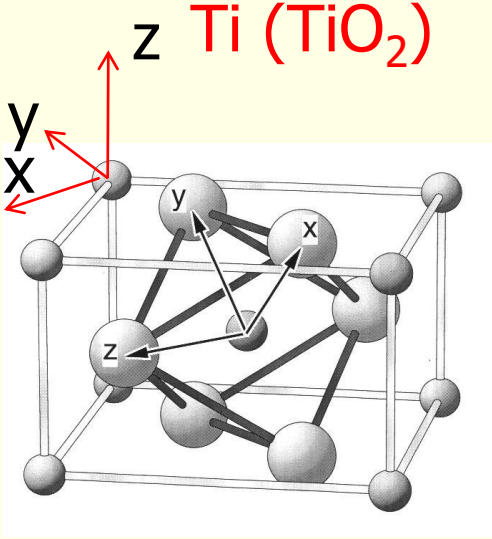
```
Title
 -0.50 0.002 1.500 0.003    EMIN, DE, EMAX, Gauss-broadening(>;de)
    3                       NUMBER OF DOS-CASES specified below
    0     1   total         atom, case=column in qtl-header, label
    1     2   Atom1-s
    2     5   Atom2-eg
```

partial charges:

- local rotation matrix:
  - *transfers z (y) into highest symmetry*
  - *reduces terms in LM series*
  - *"chemical" interpretation*
    - p_x is different from p_y

$$\begin{pmatrix} 1/\sqrt{2} & 1/\sqrt{2} & 0 \\ -1/\sqrt{2} & 1/\sqrt{2} & 0 \\ 0 & 0 & 1 \end{pmatrix}$$

Ti (TiO_2)



  - *see case.struct and case.outputs*

- x qtl (instead of x lapw2 -qtl)
  - ***f**-orbitals*
  - *qtls for **different coordinate system** (eg."octahedral" in TiO_2)*
  - *relativistic basis (**$p_{1/2}$-$p_{3/2}$** or **$d_{3/2}$-$d_{5/2}$** splitting in so calculation)*
  - *for angular dependend TELNES (ISPLIT 88, 99)*

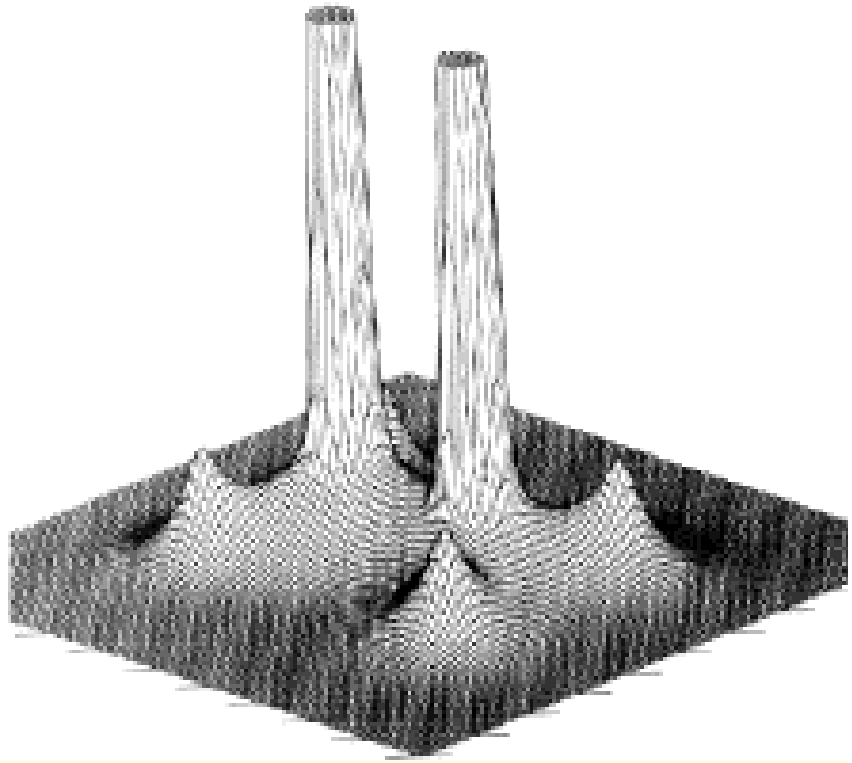
Properties with WIEN2k - I

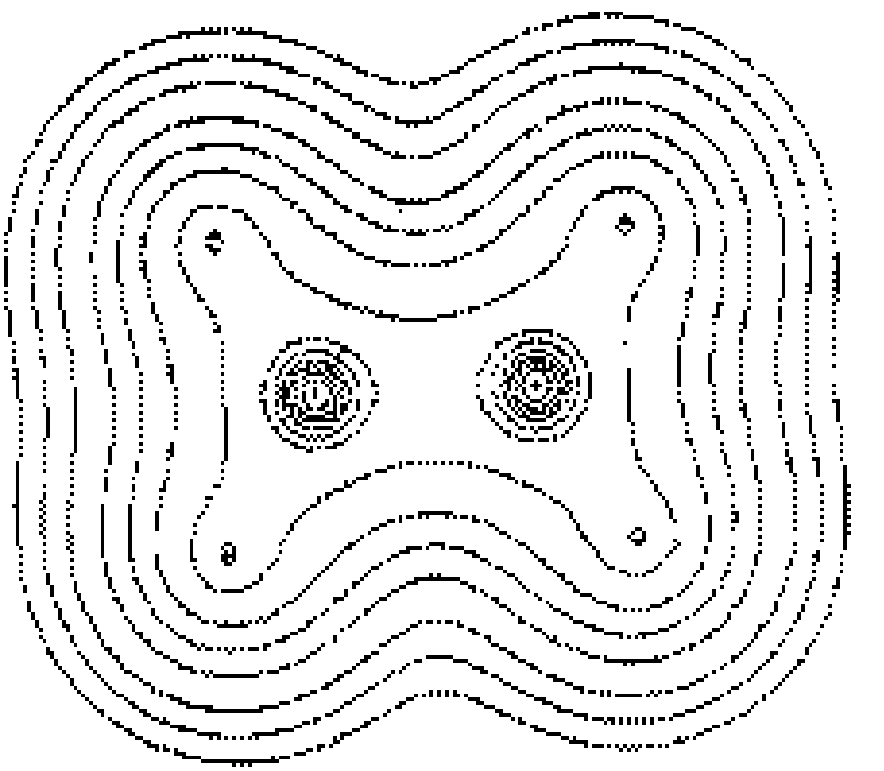
- Energy bands
  - *classification of irreducible representations*
  - *´character-plot´ (emphasize a certain band-character)*
- Density of states
  - *including partial DOS with l and m- character (eg. p_x , p_y , p_z)*
- Electron density, potential
  - *total-, valence-, difference-, spin-densities, ρ of selected states*
  - *1-D, 2D- and 3D-plots (Xcrysden)*
  - *X-ray structure factors*
  - *Bader´s atom-in-molecule analysis, critical-points, atomic basins and charges ($\nabla\rho.\vec{n}=0$)*
  - *spin+orbital magnetic moments (spin-orbit / LDA+U)*
- Hyperfine parameters
  - *hyperfine fields (contact + dipolar + orbital contribution)*
  - *Isomer shift*
  - *Electric field gradients*

Atoms in Molecules

- Theory to characterize atoms and chemical bonds from the topology of the electron density, by R.F.Bader (http://www.chemistry.mcmaster.ca/faculty/bader/aim/aim_0.html)

Electron density of C_2H_4

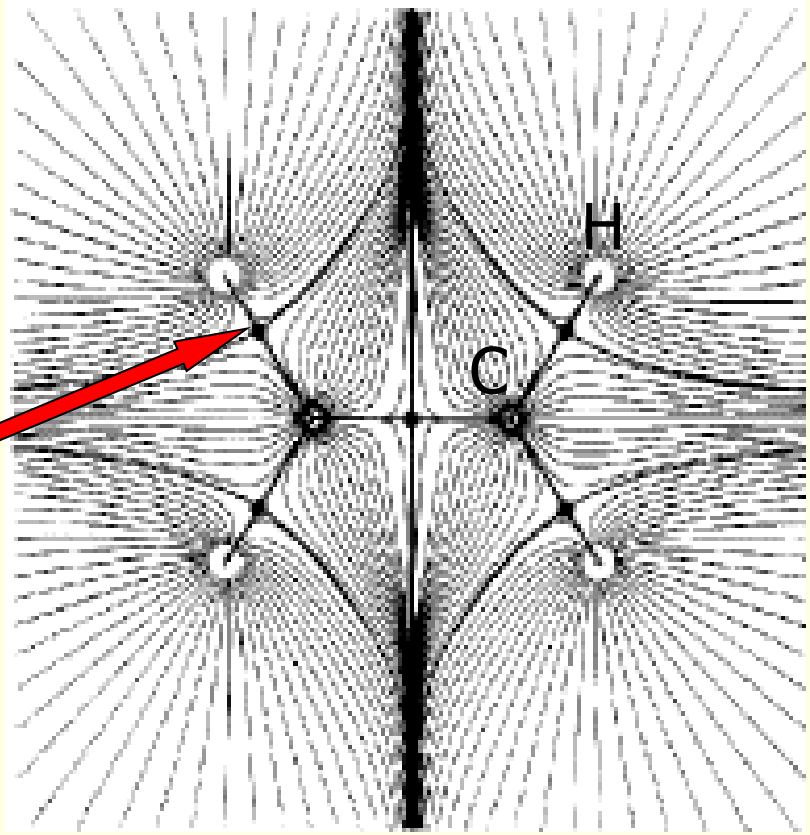




AIM-II

- Bonds are characterized by „critical points“, where $\nabla \rho = 0$
  - density maximum: (3,-3); 3 negative curvatures λ, (at nucleus or non-NM)
  - bond CP: (3,-1): 2 negative, 1 positive λ (saddle point)
    - positive (and large) Laplacian: ionic bond
    - negative Laplacian: covalent bond
  - bridge CP: (3,1)
  - cage CP: (3,3) (minimum)

(3,-1) BCP

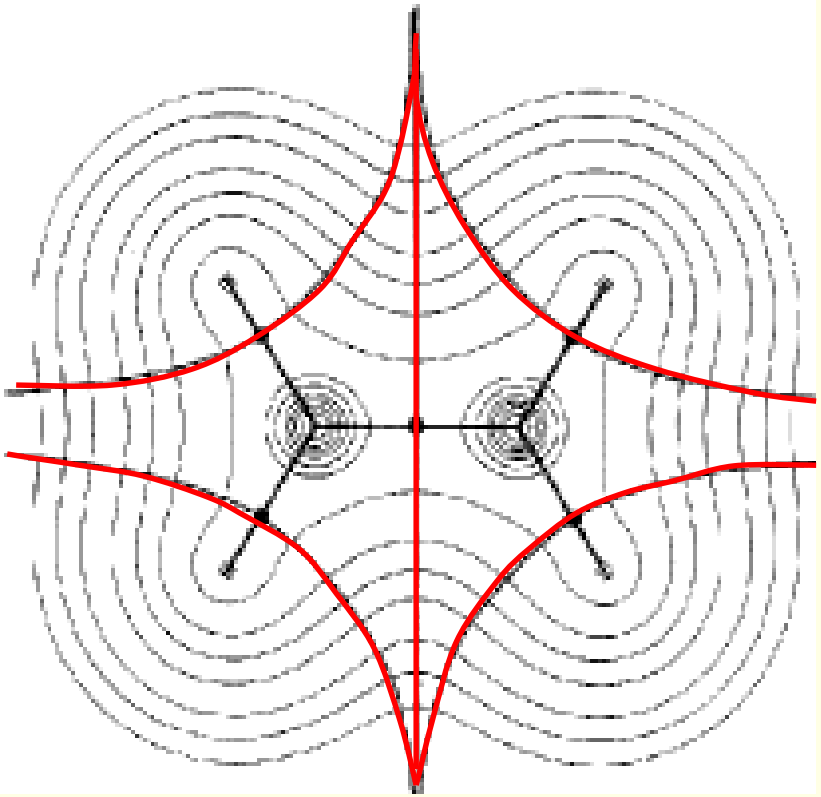


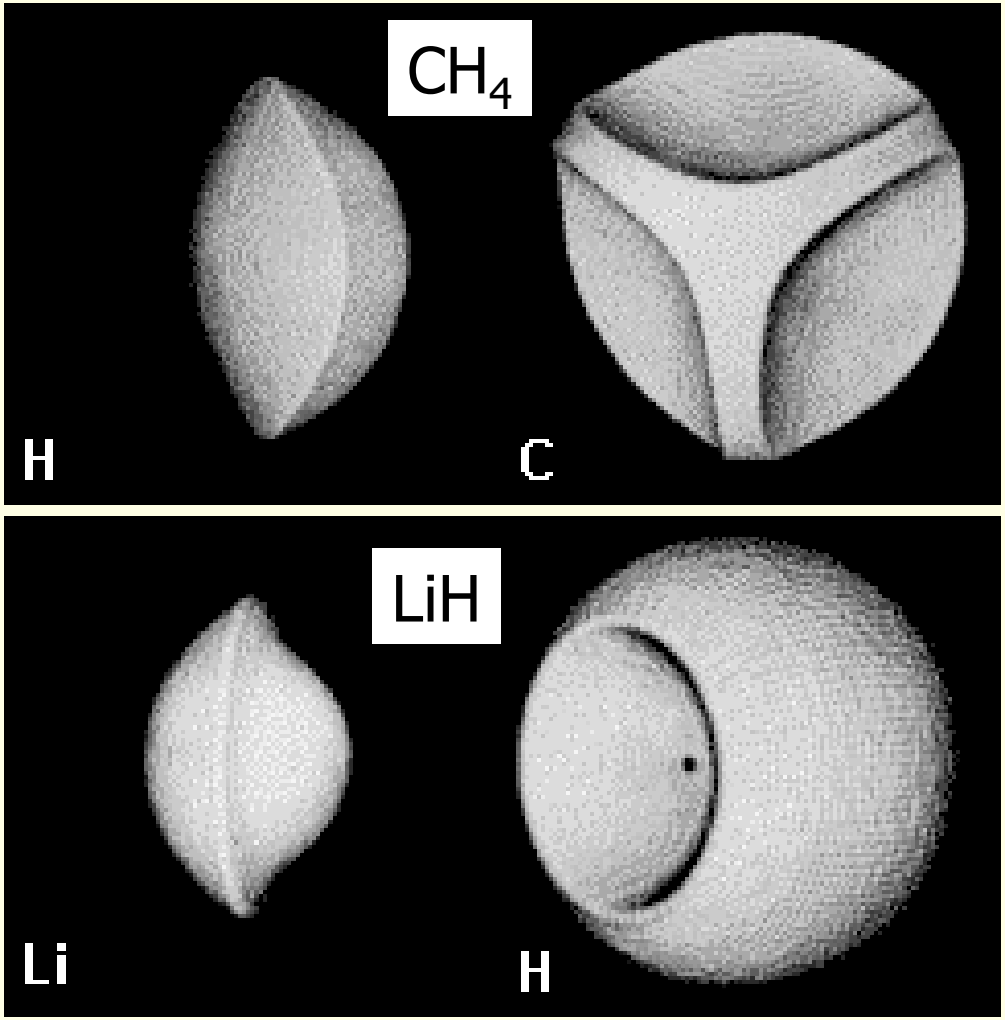
trajectories of constant $\nabla \rho$
originating at CPs in C_2H_4

AIM-III

- "Atoms" are regions within a zero-flux surface $\vec{\nabla}\rho \cdot \vec{n} = 0$

ρ of C_2H_4 with zero-flux lines defining atomic basins





AIM-IV

- 2 tools: **x aim** or **critic2** (see unsupported software)
- Bader analysis of some inorganic compounds:

	$\rho(e/A^3)$	$\Delta\rho(e/A^5)$	Q (e)
Cl_2	1.12	-6.1	-
I_2	0.48	-0.9	-
TiC	0.51	1.8	1.7
TiN	0.47	3.9	1.7
TiO	0.43	5.8	1.5
KCl	0.08	1.2	0.6

Cl_2 more covalent then I_2

more ionic, but less charge?

less ionic then TiC ?

- x aim –up: gives "unique" magnetic moments/atom

x aim

- You must have a "good" scf-density (case.clmsum)
  - *no core leakage, LMs up to L=8-10 in case.in2*

```
SURF
1                       atom in center of surface (including MULT)
20 0.0 1.570796327      theta, 20 points, from zero to pi/2
20 0.0 0.785398163      phi, from 0 to pi/4   (depends on symmetry!!)
0.07 1.0 4              step along gradient line, rmin (has reached an atom)
1.65 0.1                initial R for search, step (a.u)
3 3 3                   nshell
IRHO                    "INTEGRATE" rho
WEIT                    WEIT (surface weights are available in case.surf)
30                      30 radial points outside min(RMIN,RMT)
END
--------------------
CRIT
1                       atom around you search for critical points
ALL                     two, three, four, all    (dimers,trimers,....all=2+3)
3 3 3                   nshell
END
```

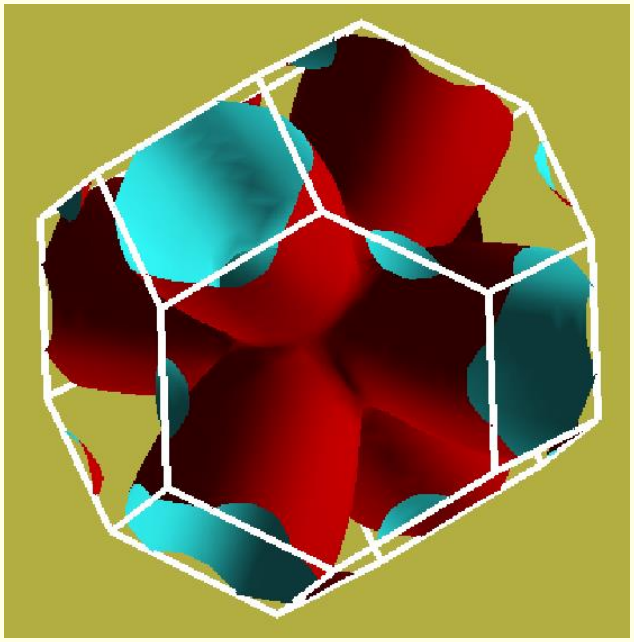
extractaim_lapw: ➔ critical_points_ang (converted units)
:PC x, y, z, λ_1, λ_2, λ_3, ch, laplacian, rho

Properties with WIEN2k - II

- Total energy and forces
  - *optimization of internal coordinates, (MD, BROYDEN)*
  - *cell parameter only via E_{tot} (no stress tensor)*
  - *elastic constants for cubic, hexagonal, and tetragonal cells*
  - *Phonons via supercells*
    - interface to PHONON (K.Parlinski) – bands, DOS, thermodynamics, neutrons
    - interface to PHONOPY (A. Togo)
      - http://www.wien2k.at/reg_user/unsupported
- Spectroscopy
  - *core level shifts*
  - *X-ray emission, absorption, electron-energy-loss (with core holes)*
    - core-valence/conduction bands including matrix elements and angular dep.
  - *optical properties (dielectric function in RPA approximation, JDOS including momentum matrix elements and Kramers-Kronig)*
- fermi surface: 2D, 3D (using XcrysDen)

Fermi surfaces

- **`xcrysden --wien_fermisurface tin.struct`**
  - choose a good k-mesh (eg. 10000 points)
  - plot the FS for all bands which cross E_F and compare to band structure



- *for 2D plots there is also a WIEN2k-tool „fsgen" (see UG)*
- *SKEAF (www.wien2k.at/reg_users/unsupported): quantum oszillations*

Properties with WIEN2k - II

- Total energy and forces
  - *optimization of internal coordinates, (MD, BROYDEN)*
  - *cell parameter only via E_{tot} (no stress tensor)*
  - *elastic constants for cubic, hexagonal, and tetragonal cells*
  - *Phonons via supercells*
    - interface to PHONON (K.Parlinski) – bands, DOS, thermodynamics, neutrons
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- Spectroscopy
  - *core level shifts*
  - *X-ray emission, absorption, electron-energy-loss (with core holes)*
    - core-valence/conduction bands including matrix elements and angular dep.
  - *optical properties (dielectric function in RPA approximation, JDOS including momentum matrix elements and Kramers-Kronig)*
- fermi surface: 2D, 3D (using XcrysDen)

Cohesive energy

$$E_{A_xB_y}^{cohes.} = E^{crystal} - xE_A^{atom} - yE_B^{atom}$$

- $E^{crystal}$: scalar-relativistic valence (or approx. SO)

- E^{atom} : LSTART: fully-relativistic➔ inconsistent description

  ➔ for heavier elements (2nd row):

  supercell with one atom in a ~30-40 bohr **distorted** FCC box (identical RMT, 1 k-point, **equivalent** RKmax, spinpolarized, iterative diagonalization)

RMT^A=2.0, RMT^B=1.8, Rkmax=7 → $RKmax^A$=7*2.0/1.8, $RKmax^B$=7

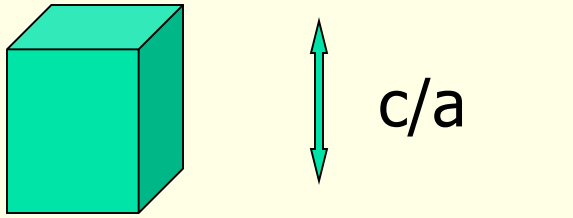
Structural optimizations:

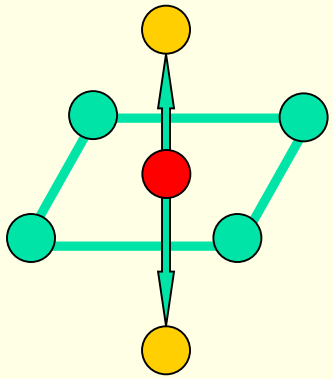
- Lattice parameters, volume, c/a ratio only via total energies:
  - *x optimize [-job "run –ec 0.000001" -save pbe]: creates a series of "struct" files + script "optimize.job"*
    - select volume or c/a, …
    - select number of cases and desired changes in volume (in % of V_0)
  - *edit optimize.job*
    - adapt to your need: change / uncomment various lines, eg.:
      - replace "run" by "runsp" or add options (-min –fc 1 –orb –i 80 –p –fc 1.0)
      - modify "save_lapw" line (with more specific names)
  - *execute optimize.job*
  - *plot (analyse) the results (eplot –a pbe)*

  - *combinations of volume and c/a are possible: 2Doptimize*
    - "x optimize" always uses case_initial.struct (if present)
    - do a "volume" optimization to create case_vol_xx.struct files
    - copy the respective case_vol_xx.struct file to case_initial.struct
    - x optimize with "c/a" for this particular volume and proceed as above.
  - *optimize_abc_lapw: for 2D (a,c) or 3D (a,b,c) optimization*

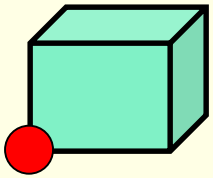
Symmetry:

- WIEN „preserves" symmetry:
  - *c/a optimization of „cubic" TiC:*
    - change c lattice parameter in TiC.struct (tetragonal distortion, #sym.op=0)
    - init_lapw
    - change c back to cubic
    - x optimize …
  - *„Jahn-Teller" distortion:*
    - when you start with a perfect octahedra, you will never get any distortion
    - ➔start with slightly distorted positions

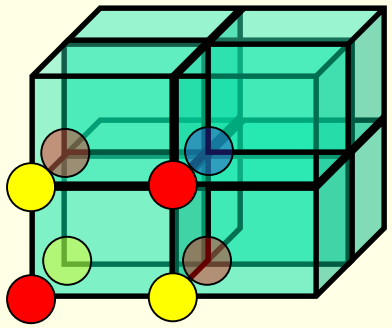




Supercells (impurities, vacancies, alloys)



2x2x2 = 8 atoms

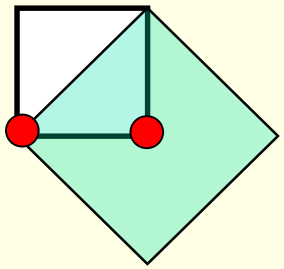


(0,0,0) P➔ 8 atoms (0,0,0) (.5,0,0) (.5,.5,0) (.5,.5,.5) (0,.5,0) (.5,0,.5) (0,0,.5) (0,.5,.5)

B➔ 4 atoms	yes	yes	no	no
F➔ 2 atoms	yes	no	no	yes

4x4x4 supercells: P (64), B (32), F (16) atoms

$\sqrt{2}\,x\sqrt{2}$ supercells (1 → 2 atoms)



Supercells

- Program „supercell“:
  - *start with „small“ struct file*
  - *specify number of repetitions in x,y,z (only integers, e.g. 2x2x1)*
  - *specify P, B or F lattice*
  - *add „vacuum“ for surface slabs (only (001) indexed surfaces)*
  - *shift all atoms in cell*
- You must break symmetry !!! (otherwise sgroup will restore your original struct file)
  - *replace (impurities, vacancies) or*
  - *displace (phonons) or*
  - *label at least 1 atom (core-holes, specific magnetic order; change "Fe" to "Fe1"; this tells the symmetry-programs that Fe1 is NOT a Fe atom!!)*
- „supercell“ works only along unit-cell axes!!!

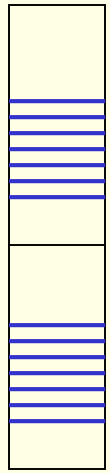
Structeditor (by R.Laskowski)

- requires octave (matlab) and xcrysden (visualization)
- allows complex operations on struct-files

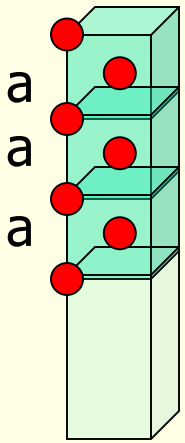
```
octave
s=loadstruct("GaN.struct")
# make an orthorhombic supercell and visualize it
a=[1 0 0; 1 1 0; 0 0 2]
sout=makesupercell (s,a);
showstruct(sout);
# save it as test.struct
savestruct (sout,"test.struct");
# get help on all commands
helpstruct
```

Surfaces

- 2D-slabs with finite number of layers with „vacuum“ in 3rd dimension



bcc (001) 7 layers:



(0 0 6z)
(.5 .5 5z)
(0 0 4z)
(.5 .5 3z)
(0 0 2z)
(.5 .5 z)
(0 0 0)

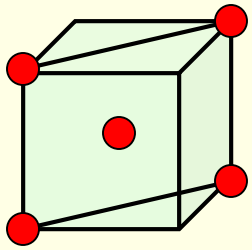
shift to ➔ inversion

(.5 .5 +/-3z)
(0 0 +/-2z)
(.5 .5 +/-z)
(0 0 0)

with lattice parameters:
a, a, c=(3a+15-20bohr vacuum)

z= a/2c

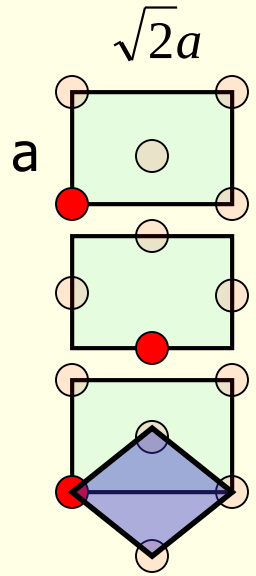
bcc (110):



orthorhombic CXY-lattice: a, $\sqrt{2}a$, c

(0 0 0)
(0 .5 +/-z)
(0 0 +/-2z)

z=a/$\sqrt{2}a$ c



Total energies and atomic forces

(Yu et al.; Kohler et al.)

- Total Energy:
  - *Electrostatic energy*
  - *Kinetic energy*
  - *XC-energy*

$$U[\rho] = \tfrac{1}{2}\int d^3\vec{r}\ \rho(\vec{r})V_{es}(\vec{r}) + \tfrac{1}{2}\sum_\alpha Z_\alpha V_{es}^\alpha(\vec{r})$$

$$T[\rho] = \sum_i n_i\varepsilon_i - \int d^3\vec{r}\ \rho(\vec{r})V_{eff}(\vec{r})$$

$$E_{xc}[\rho] = \int d^3\vec{r}\ \rho(\vec{r})\varepsilon_{xc}(\vec{r})$$

- Force on atom α:

$$\vec{F}^\alpha = \frac{-dE_{tot}}{d\vec{R}_\alpha} = F_{HF}^\alpha + F_{core}^\alpha + F_{val}^\alpha$$

  - *Hellmann-Feynman-force*
  - *Pulay corrections*
    - Core
    - Valence
- expensive, contains a summation of matrix elements over all occupied states

$$F_{HF}^\alpha = Z_\alpha \sum_{m=-1}^{1} \lim_{r_\alpha\to 0} \frac{V_{1m}^{es}(r_\alpha)}{r_\alpha} \nabla_\alpha\left[r_\alpha Y_{1m}(\hat{r})\right]$$

$$F_{core}^\alpha = -\int \rho_{core}(r)\nabla_\alpha V_{eff}(r)\, d\vec{r}$$

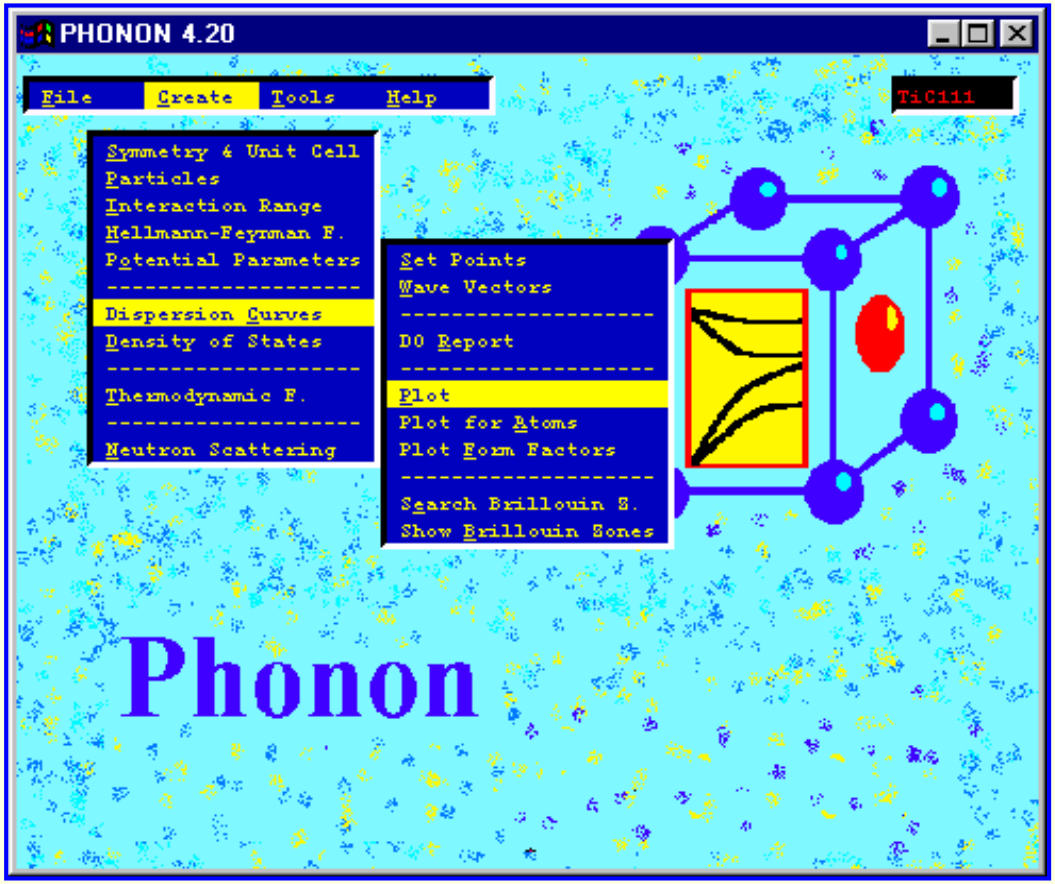
$$F_{val}^\alpha = \int_\alpha V_{eff}(r)\nabla_\alpha\rho_{val}(r)\, d\vec{r} + \sum_{k,i} n_i \sum_{K,K'} c_i^*(K')c_i(K)\times$$

$$\left[(K^2-\varepsilon_i)\oint \phi_{K'}^*(r)\phi_K(r)\, dS_\alpha - i(K-K')\left\langle\phi_{K'}\middle|H-\varepsilon_i\middle|\phi_K\right\rangle_\alpha\right]$$

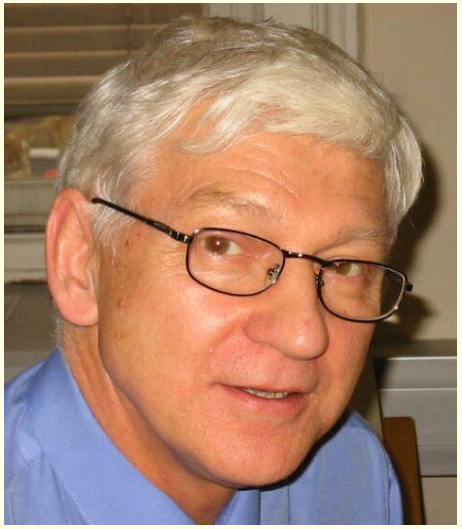
Optimization of internal parameters using "forces"

- Forces only for "free" structural parameters:
  - *NaCl: (0,0,0), (0.5,0.5,0.5) : all positions fixed by symmetry*
  - *TiO_2: Ti (0,0,0), O (u,u,0): one free parameter (u,x,y,z)*
- Forces are only calculated when using "-fc":
  - *run_lapw –fc 1.0 (mRy/bohr)*
    - grep :fgl002 case.scf
      - 200. partial
      - -130. partial
      - 140. partial
      - 135 partial — only $F_{HF} + F_{core}$
      - 120 partial
      - 122 partial — forces converging
      - 121 partial — → changes "TOT" to "FOR" in case.in2
      - -12.3 **total** — $F_{HF} + F_{core} + F_{val}$, only this last number is correct
- Forces are useful for
  - *structural optimization (of internal atomic positions)*
  - *phonons*

Calculations of Phonons: The Direct Method



WIEN2k + Phonon
WIEN2k + Phonopy
WIEN2k + Phon



Copyright by K.Parlinski

Phonopy: https://phonopy.github.io/phonopy/ by A.Togo. Most used phonon code, free.
Phonon: https://computingformaterials.com by K.Parlinski. Graphical user interface, expensive
Phon: http://www.homepages.ucl.ac.uk/~ucfbdxa/phon/ by D.Alfe. free

See http://www.wien2k.at/unsupported)

THEORY OF DIRECT METHOD

System energy E (at $T = 0$) as a function of atomic positions $\mathbf{R}(\mathbf{n}, \mu)$ is

$$E(\mathbf{R}(\mathbf{n},\mu),..\mathbf{R}(\mathbf{m},\nu),\ldots) = E_o + \frac{1}{2}\sum_{\mathbf{n},\mu,\mathbf{m},\nu} \Phi(\mathbf{n},\mu,\mathbf{m},\nu)\mathbf{U}(\mathbf{n},\mu)\mathbf{U}(\mathbf{m},\nu)$$

where the *force constant matrix* are

$$\Phi_{i,j}(\mathbf{n},\mu,\mathbf{m},\nu) = \frac{\partial^2 E}{\partial \mathbf{R}_i(\mathbf{n},\mu)\partial \mathbf{R}_j(\mathbf{m},\nu)}\Big]_o$$

is defined at $\frac{\partial E}{\partial \mathbf{R}_i(\mathbf{n},\mu)}\,|_o = 0$.

The dynamical matrix is defined as

$$\mathbf{D}(\mathbf{k};\mu,\nu) = \frac{1}{\sqrt{M_\mu M_\nu}}\sum_{\mathbf{m}} \Phi(0,\mu;\mathbf{m},\nu)\, exp\{-2\pi i\mathbf{k}\cdot[\mathbf{R}(0,\mu) - \mathbf{R}(\mathbf{m},\nu)]\}$$

$\mathbf{m}$ runs over *all* atoms. Diagonalization of the dynamical matrix

$$\omega^2(\mathbf{k},j)\mathbf{e}(\mathbf{k},j) = \mathbf{D}(\mathbf{k})\mathbf{e}(\mathbf{k},j)$$

gives phonon frequencies $\omega^2(\mathbf{k}, j)$ and polarization vectors $\mathbf{e}(\mathbf{k}, j)$.

Any atomic displacement $\mathbf{U}(\mathbf{m}, \nu)$ generates forces

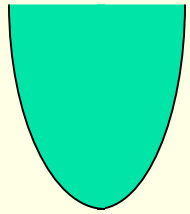
$$\mathbf{F}(\mathbf{n},\mu) = -\partial E/\partial \mathbf{R}(\mathbf{n},\mu)$$

on all other atoms. Hence

$$F_i(\mathbf{n},\mu) = -\Sigma_{\mathbf{m},\nu,j}\,\Phi_{i,j}(\mathbf{n},\mu,\mathbf{m},\nu)U_j(\mathbf{m},\nu)$$

Master equation of direct method.

$$V = \frac{1}{2}kx^2$$



n,m: cells

μ,ν: atoms

CUMMULANT FORCE CONSTANTS

Displace an atom by $\mathbf{U}(\mathbf{m}, \nu)$

$$F_i(\mathbf{n}, \mu) = -\sum_{\mathbf{L}} \Phi_{i,j}(\mathbf{n}, \mu, \mathbf{m} + \mathbf{L}), \nu) U_j(\mathbf{m}, \nu)$$

$\mathbf{L} = (L_a, L_b, L_c)$ are the indices of supercell lattice constants.

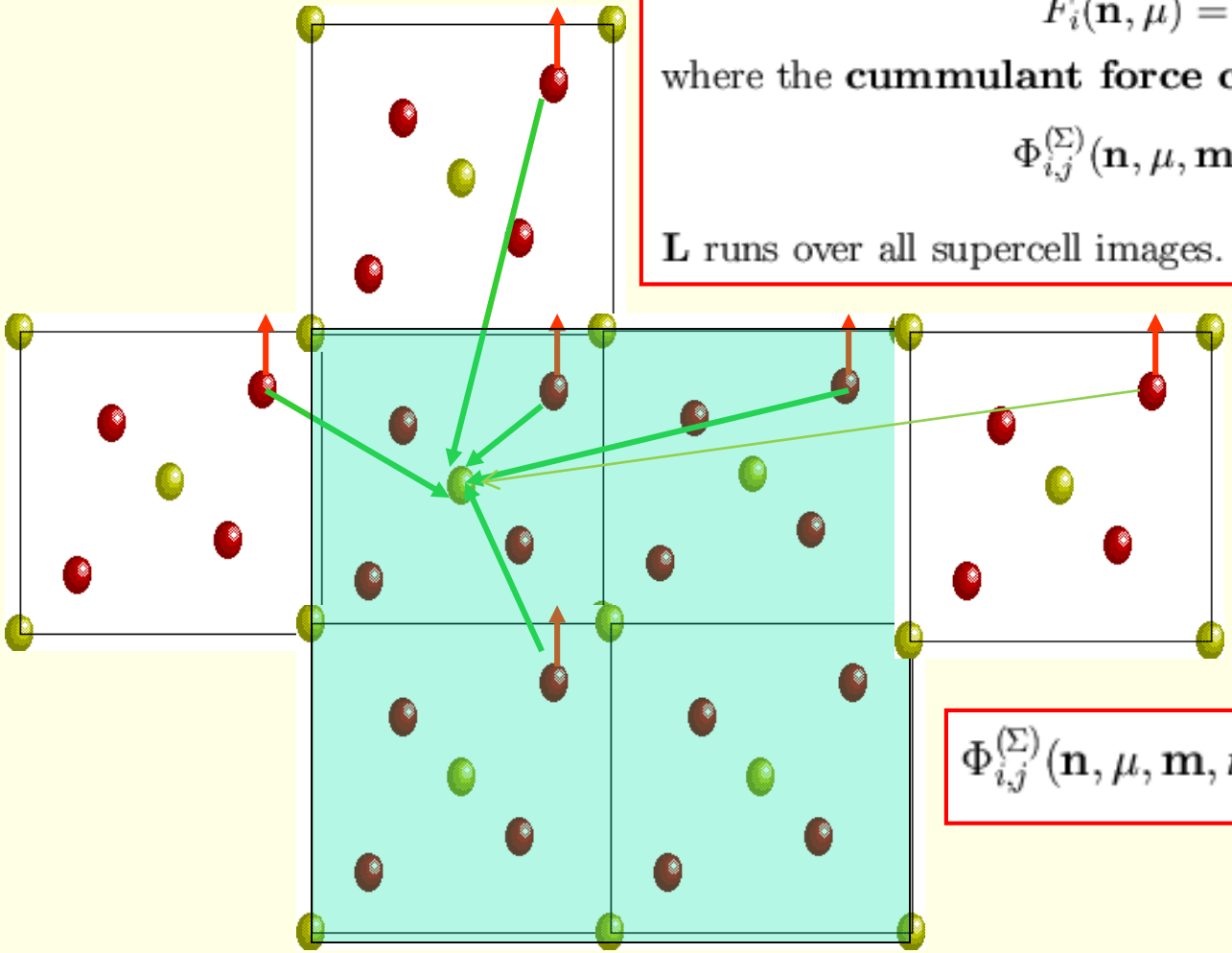
or

$$F_i(\mathbf{n}, \mu) = -\Phi^{(\Sigma)}_{i,j}(\mathbf{n}, \mu, \mathbf{m}, \nu) U_j(\mathbf{m}, \nu)$$

where the **cummulant force constant** is

$$\Phi^{(\Sigma)}_{i,j}(\mathbf{n}, \mu, \mathbf{m}, \nu) = \sum_{\mathbf{L}} \Phi_{i,j}(\mathbf{n}, \mu, \mathbf{m} + \mathbf{L}, \nu)$$

$\mathbf{L}$ runs over all supercell images.



$$\Phi^{(\Sigma)}_{i,j}(\mathbf{n}, \mu, \mathbf{m}, \nu) = \sum_{\mathbf{L}} \Phi_{i,j}(\mathbf{n}, \mu, \mathbf{m} + \mathbf{L}, \nu)$$

Supercell dynamical matrix. Exact wave vectors.

Conventional dynamical matrix:

$$\mathbf{D}(\mathbf{k};\mu,\nu) = \frac{1}{\sqrt{M_\mu M_\nu}} \sum_{\mathbf{m}} \Phi(0,\mu;\mathbf{m},\nu)\, exp\{-2\pi i \mathbf{k}\cdot[\mathbf{R}(0,\mu)-\mathbf{R}(\mathbf{m},\nu)]\}$$

Supercell dynamical matrix:

$$\mathbf{D}^{(SC)}(\mathbf{k};\mu,\nu) = \frac{1}{\sqrt{M_\mu M_\nu}} \sum_{\mathbf{m}\in SC} \Phi^{(SC)}(0,\mu;\mathbf{m},\nu)\, exp\{-2\pi i \mathbf{k}\cdot[\mathbf{R}(0,\mu)-\mathbf{R}(\mathbf{m},\nu)]\}$$

These two matrices are equal if

$$\mathbf{D}^{(SC)}(\mathbf{k};\mu,\nu) = \mathbf{D}(\mathbf{k};\mu,\nu)$$

- **interaction range** is confined **to interior** of supercell (supercell is big enough)
- wave vector is **commensurate with the supercell** and fulfils the condition (independent of interaction range):

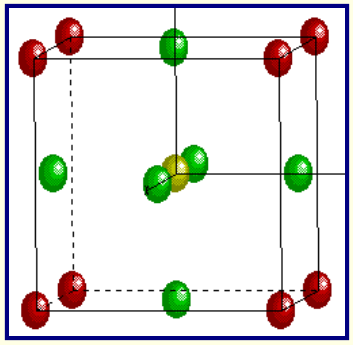
$$exp\{-2\pi i \mathbf{k}_s \cdot \mathbf{L}\} = 1$$

At wave vectors $\mathbf{k}_s$ the phonon frequencies are "exact", provided the **supercell contains the complete list of neighbors.**

Wave vectors $\mathbf{k}_s$ are commensurate with the supercell size.

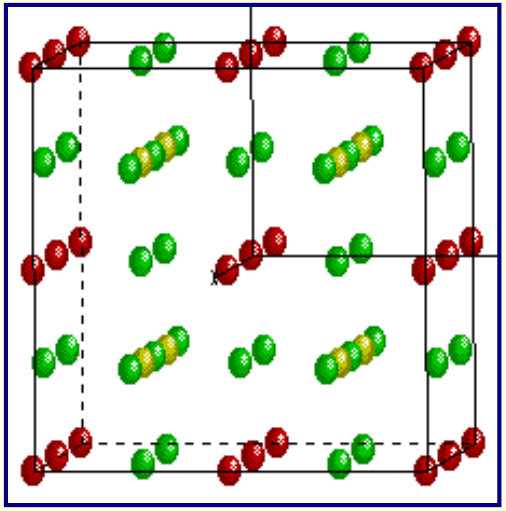
Exact wave vectors

1x1x1



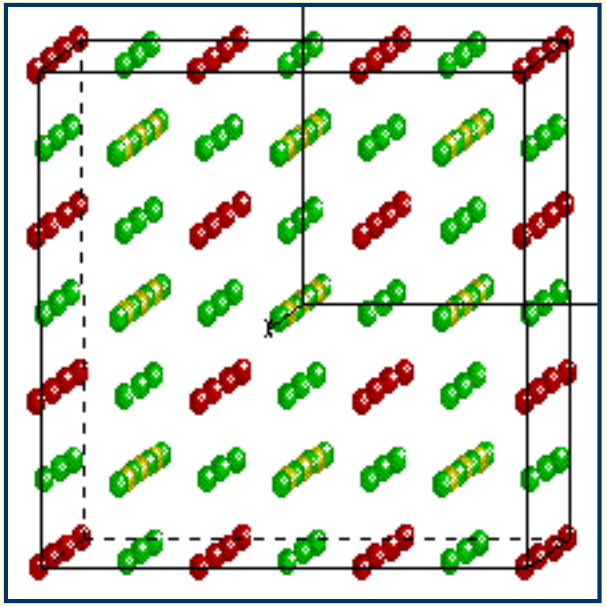
Exact: Γ

2x2x2

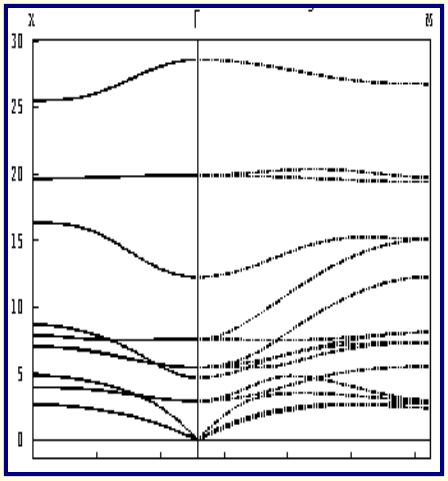


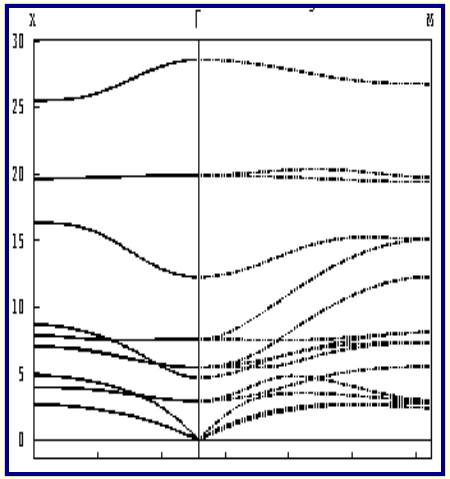
Exact: Γ, X, M, R

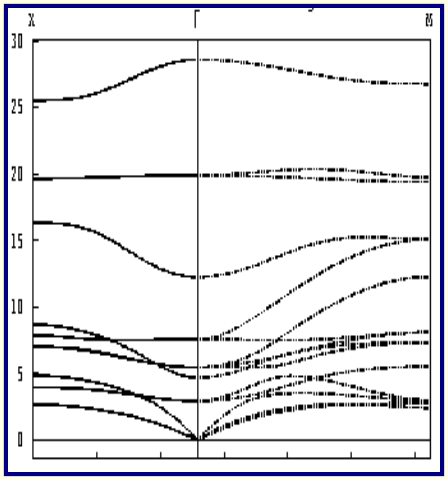
3x3x3



Exact: Γ X Γ M ↓ Γ ↓



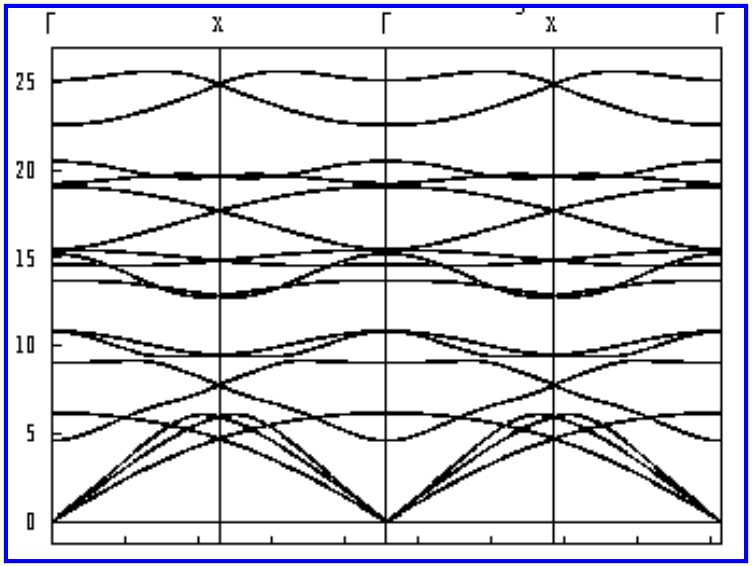




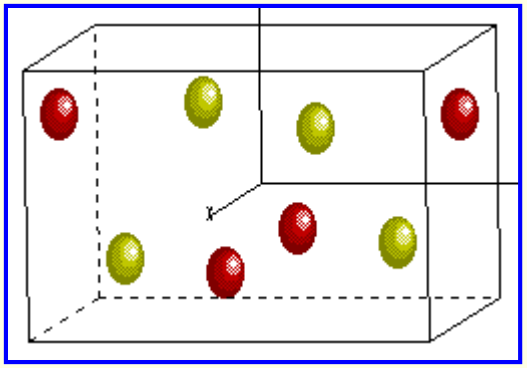
Phonon dispersions + density of states

Frequency

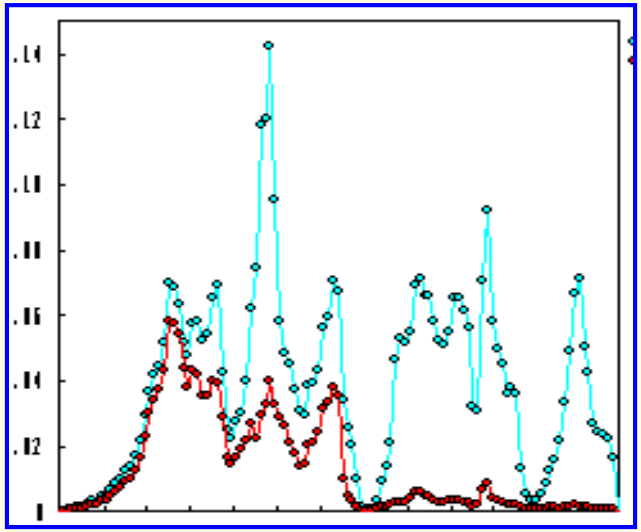
ω



GeO_2 P4_2/mnm

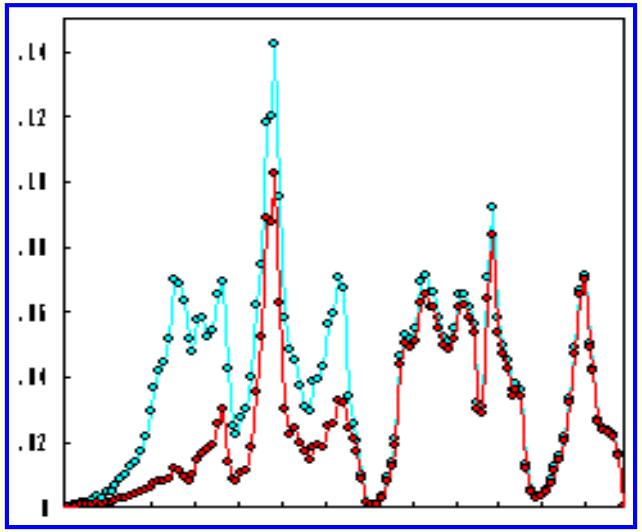


Total + Germanium



ω

Total + Oxygen



ω

Thermodynamic functions of phonon vibrations

Internal energy:

$$E = \frac{1}{2} r \int_0^\infty d\omega\, g(\omega)\, (\hbar\omega)\, coth\left(\frac{\hbar\omega}{2k_B T}\right)$$

Free energy:

$$F = r k_B T \int_0^\infty d\omega\, g(\omega)\, ln\left[2 sinh\left(\frac{\hbar\omega}{2k_B T}\right)\right]$$

Entropy:

$$S = r k_B \int_0^\infty d\omega\, g(\omega) \left\{\left(\frac{\hbar\omega}{2k_B T}\right)\left[coth\left(\frac{\hbar\omega}{2k_B T}\right) - 1\right] - ln\left[1 - exp\left(-\frac{\hbar\omega}{k_B T}\right)\right]\right\} \quad (79)$$

Heat capacity C_v:

$$C = r k_B \int_0^\infty d\omega\, g(\omega) \left(\frac{\hbar\omega}{k_B T}\right)^2 \frac{exp\left(\frac{\hbar\omega}{k_B T}\right)}{\left[exp\left(\frac{\hbar\omega}{k_B T}\right) - 1\right]^2}$$

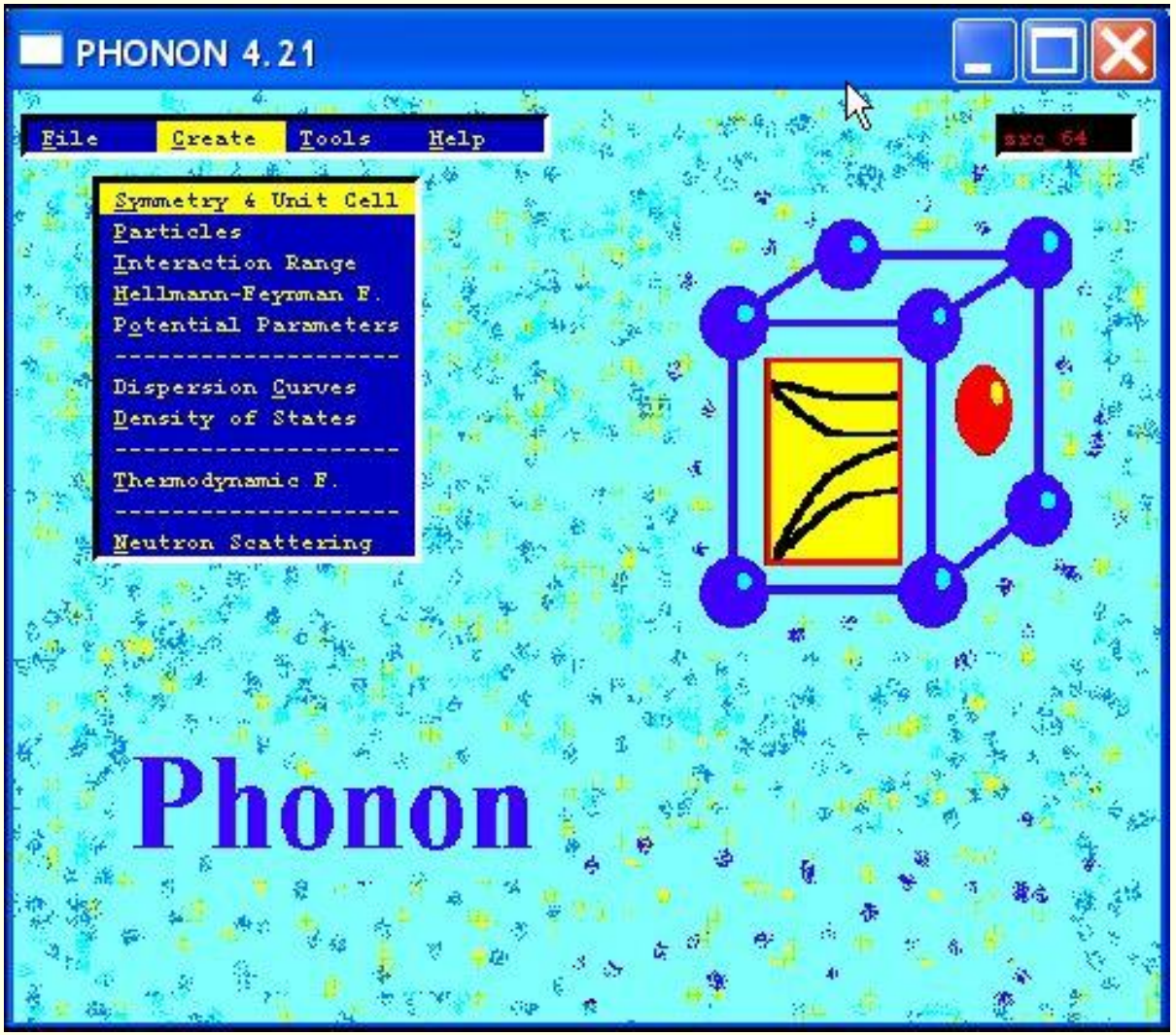
Thermal displacements:

$$B_{ij}(\mu) = < U_i(\mu)\, U_j(\mu) >$$

$$B_{il}(\mu) = \frac{\hbar r}{2M_\mu} \int_0^\infty d\omega\, g_{il,\mu}(\omega) \frac{1}{\omega} coth\left(\frac{\hbar\omega}{2k_B T}\right)$$

PHONON-I

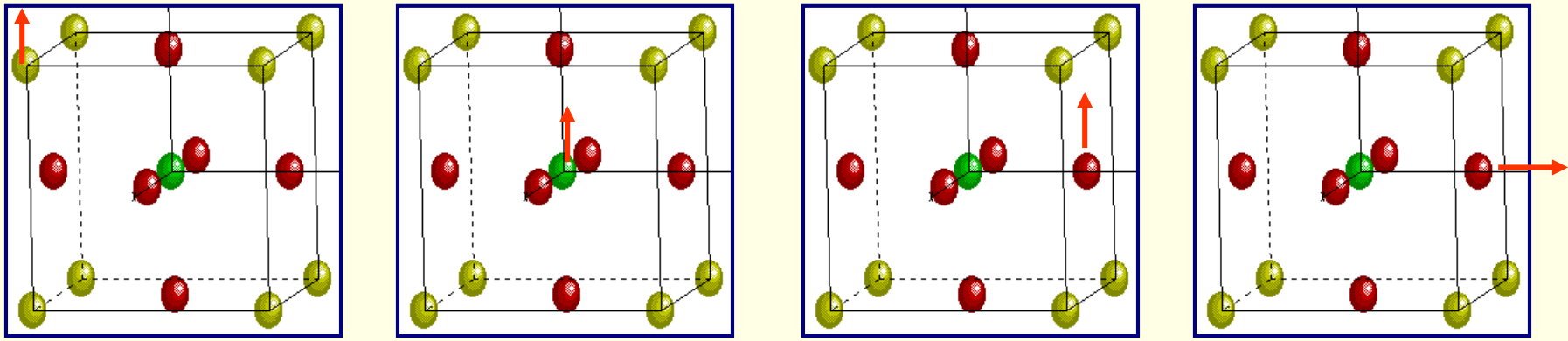
- PHONON
  - *by K.Parlinski (Crakow)*
  - *Linux or MS-windows*
  - *uses a „direct" method to calculate Force-constants with the help of an ab initio program*
  - *with these Force-constants phonons at arbitrary k-points can be obtained*
- Define your spacegroup
- Define all atoms



https://computingformaterials.com

Phonons:

- *selects symmetry adapted atomic displacements (4 displacements in cubic perovskites)*



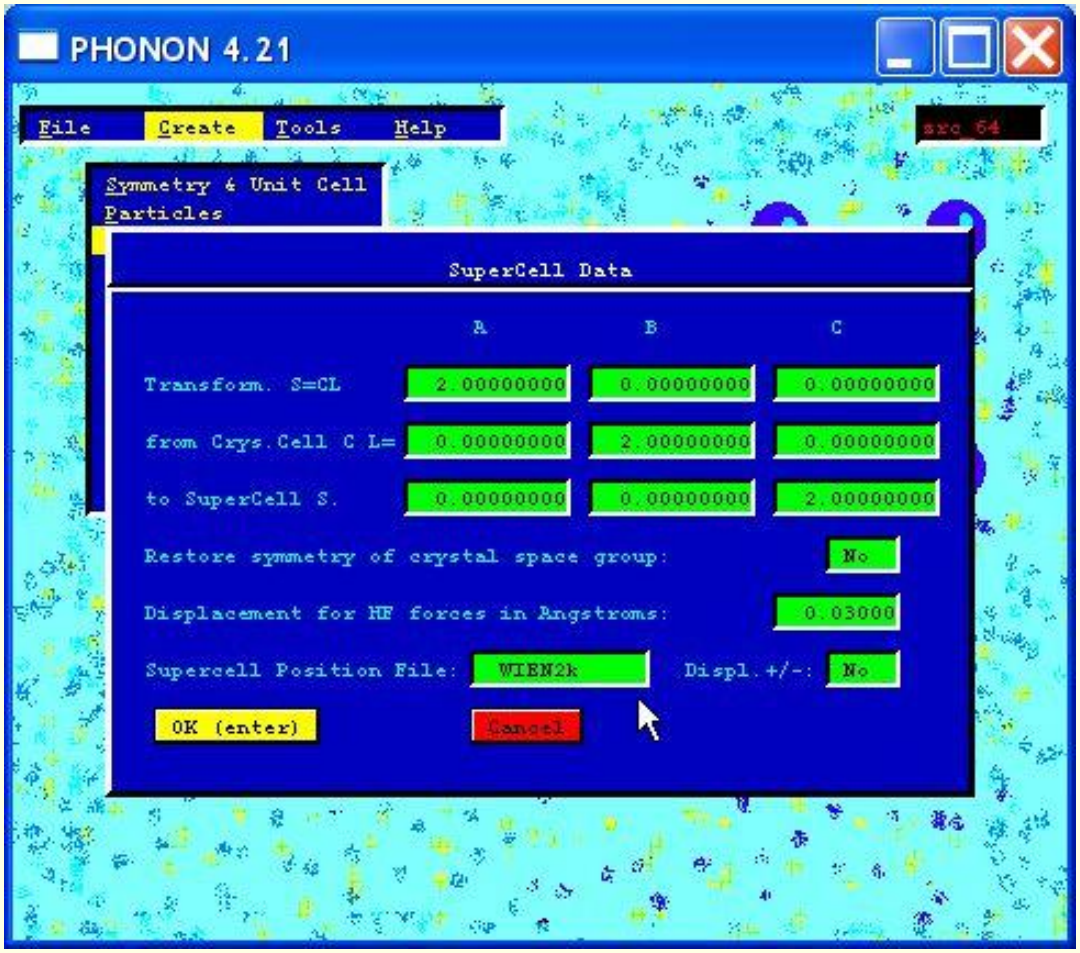
(Displacement pattern for cubic perovskite)

- *select a supercell: (eg. 2x2x2 atom P-type cell)*
- *calculate all forces for these displacements with high accuracy(WIEN2k)*

- → *force constants between all atoms in the supercell*
- → *dynamical matrix for arbitrary q-vectors*
- → *phonon-dispersion ("bandstructure") using PHONON (K.Parlinski)*

PHONON-II

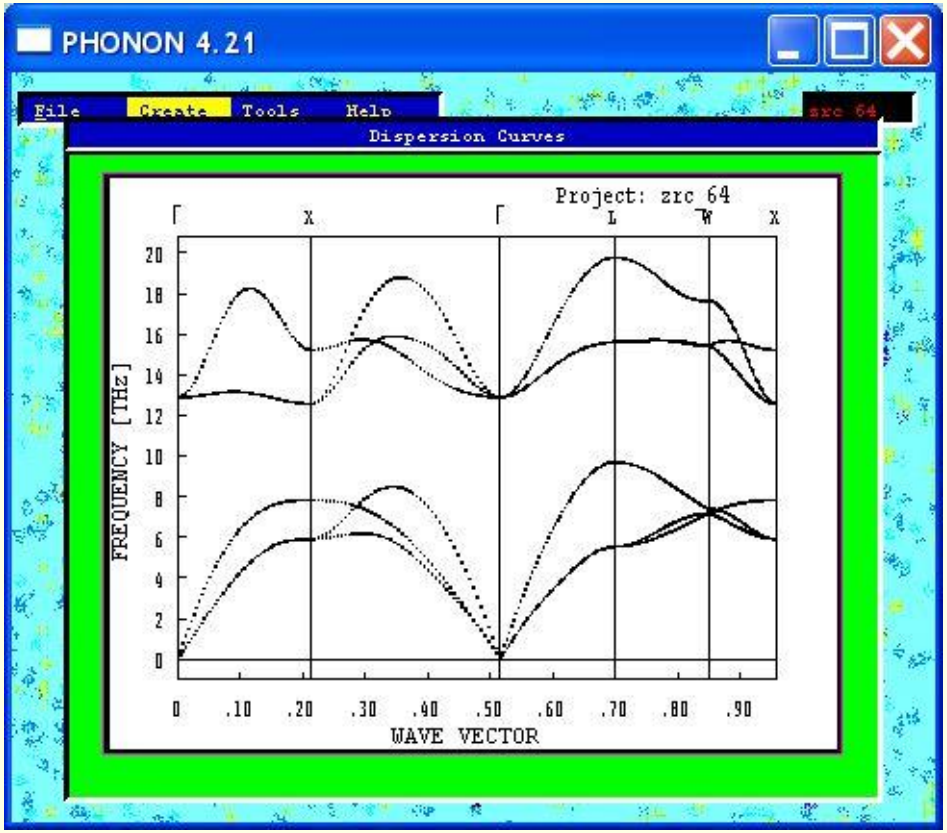
- Define an interaction range (supercell)
  - *create displacement file*
  - *transfer case.d45 to Unix*
- Calculate forces for all required displacements
  - *init_phonon_lapw*
    - for each displacement a case_XX.struct file is generated in an extra directory
    - runs nn and lets you define RMT values like:
      - 1.85 1-16



- init_lapw: either without symmetry (and then copies this setup to all case_XX)
  or with symmetry (must run init_lapw for all case_XX) (Do NOT use SGROUP)
- run_phonon: run_lapw –fc 0.1 –i 40 for each case_XX

PHONON-III

- analyze_phonon_lapw
  - *reads the forces of the scf runs*
  - *generates „Hellman-Feynman" file case.dat and a „symmetrized HF-file case.dsy (when you have displacements in both directions)*
    - check quality of forces:
    - sum F_x should be small (0)
    - abs(F_x) should be similar for +/- displacements
- transfer case.dat (dsy) to Windows
- Import HF files to PHONON
- Calculate force constants
- Calculate phonons, analyze phonons eigenmodes, thermodynamic functions



Applications:

- phonon frequencies (compare with IR, raman, neutrons)
- identify dynamically unstable structures, describe phase transitions, find more stable (low T) phases.
- free energies at T>0; quasiharmonic approximation

Pyrochlore structure of $Y_2Nb_2O_7$: strong phonon instabilities → phase transition